



U.S. DEPARTMENT OF
ENERGY

Nuclear Energy

VHTR Materials Overview

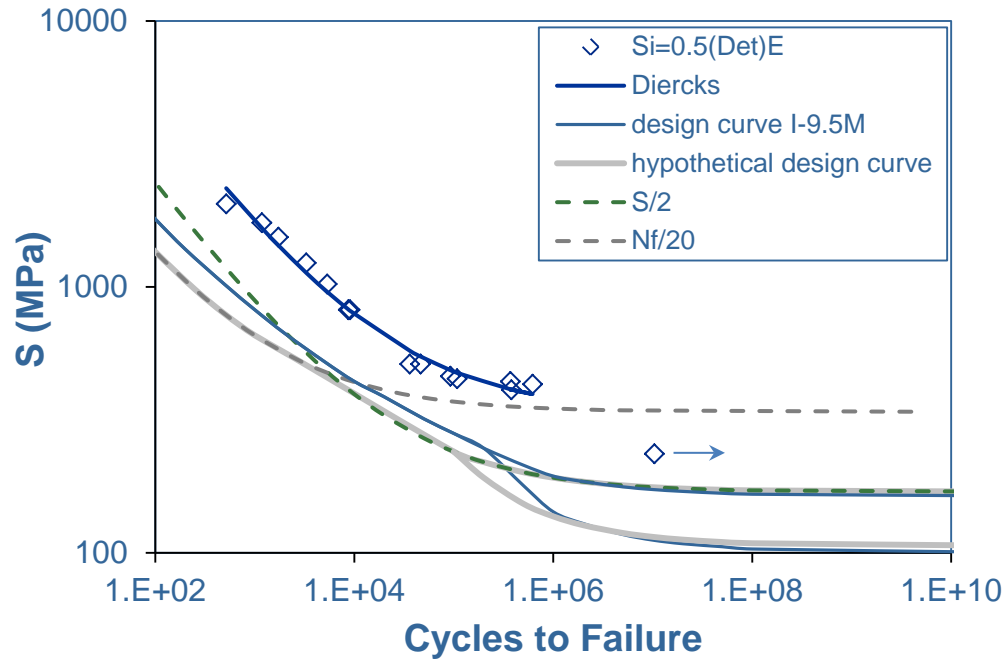
Richard Wright
Idaho National Laboratory

**DOE-NE Materials Crosscut Coordination Meeting
Tuesday July 30, 2013**

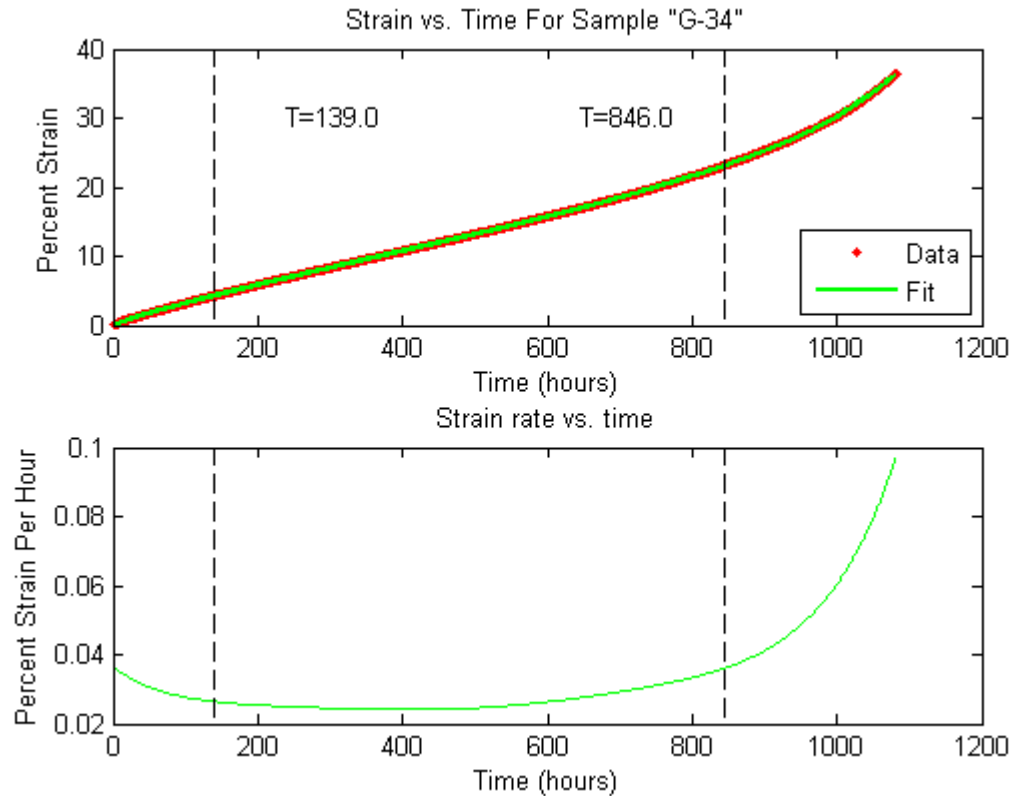
High Temperature Metals Overview

- **Focus of the program is characterization of Alloy 617 behavior and high temperature design methods for ASME Code qualification**
- **ASME Task Group on Alloy 617 Qualification has been established**
 - Two part activity
 - *Subsection NB Below 427°C data nearly complete – fatigue design curve remaining to be determined*
 - *Subsection NH above 427°C significant ongoing elevated temperature testing*
- **Support from NGNP Program, Small Modular Reactor Program and NEUP**

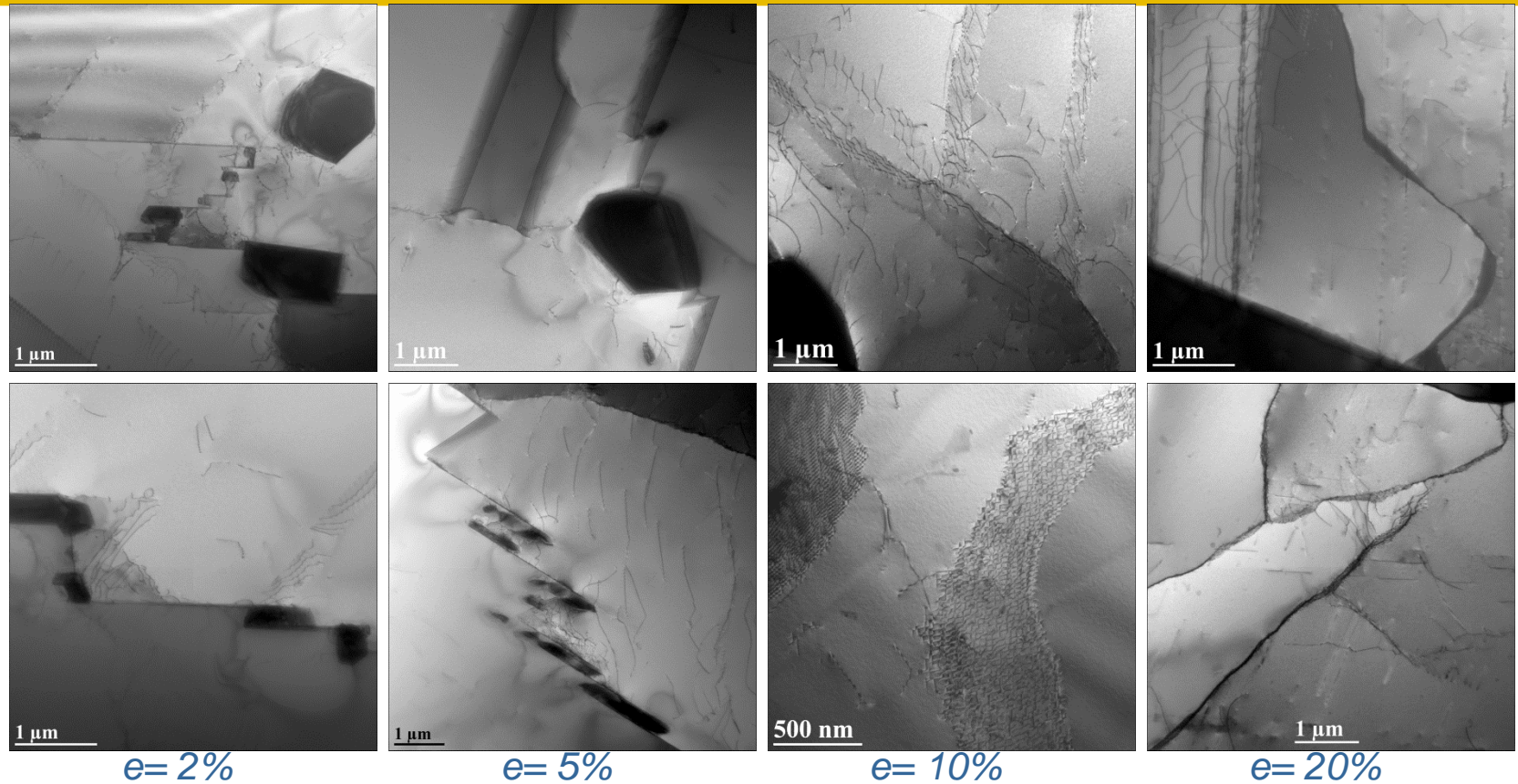
Preliminary Alloy 617 Fatigue Design Curve 427°C



Creep Curve for Alloy 617 at 800°C



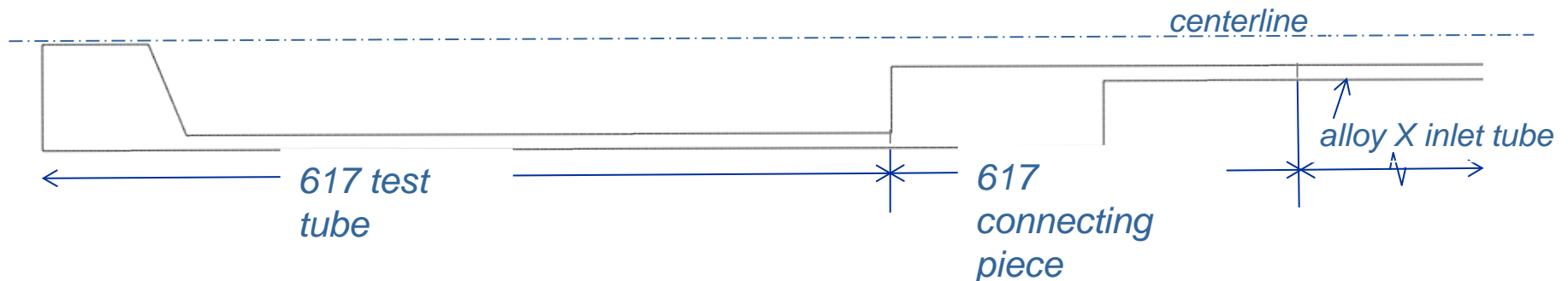
Microstructure – TEM, 1000°C, 20 MPa



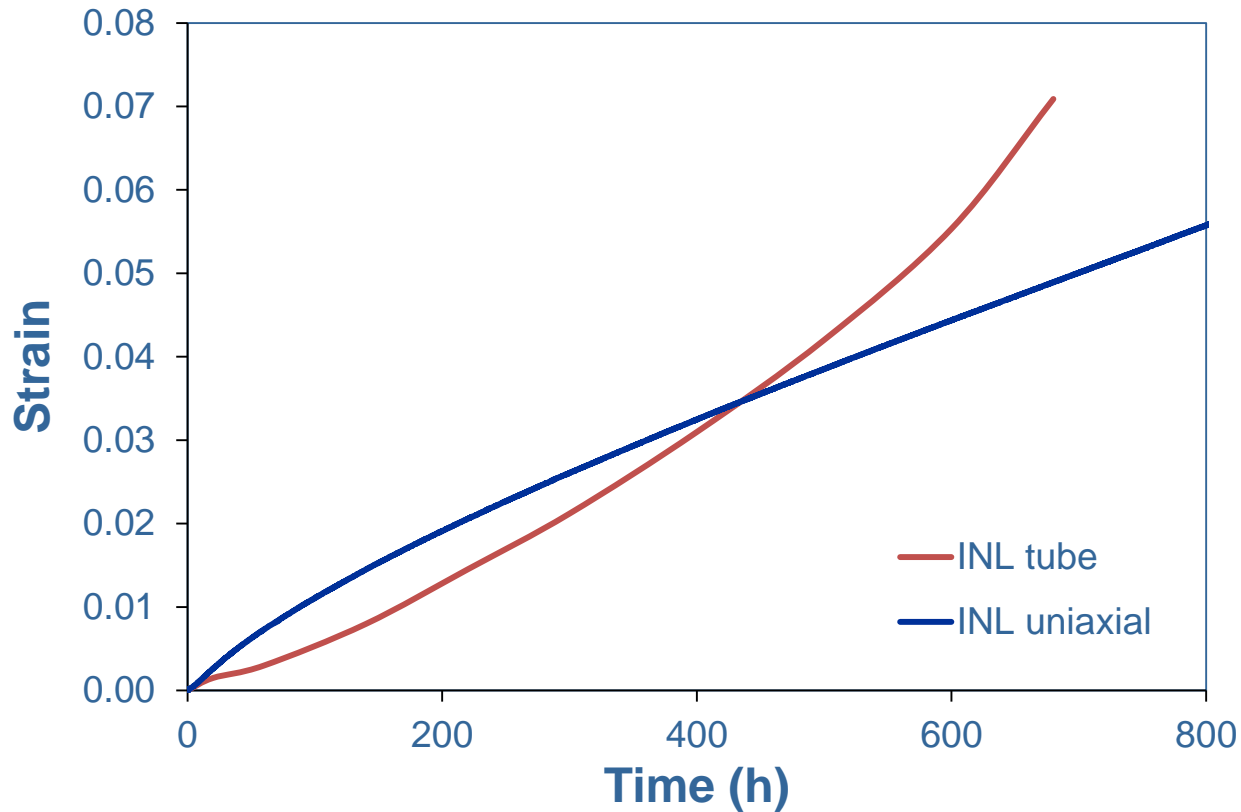
- *Minimal substructure development below 5% creep strain*
- *Extensive substructure development at 10 and 20% creep strain*
- *No porosity found during extensive examination of numerous samples*

FEM Model Pressurized Tube

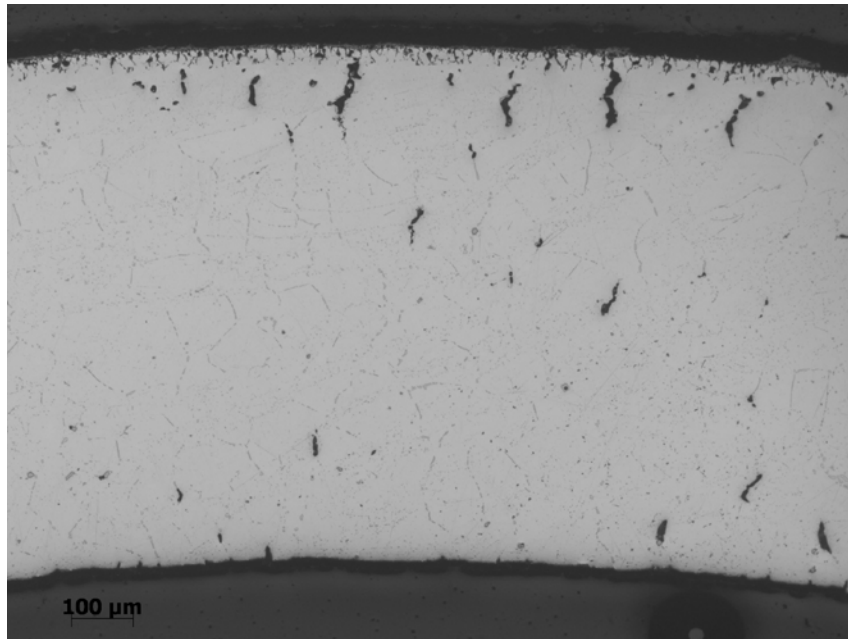
- The model was axisymmetric, so only a half cross-section of the set-up is shown
- The test tube is 12mmOD with 1 mm thick walls and 50 mm long
- 1000 psi internal applied pressure
- 950°C
- Welds were not modeled



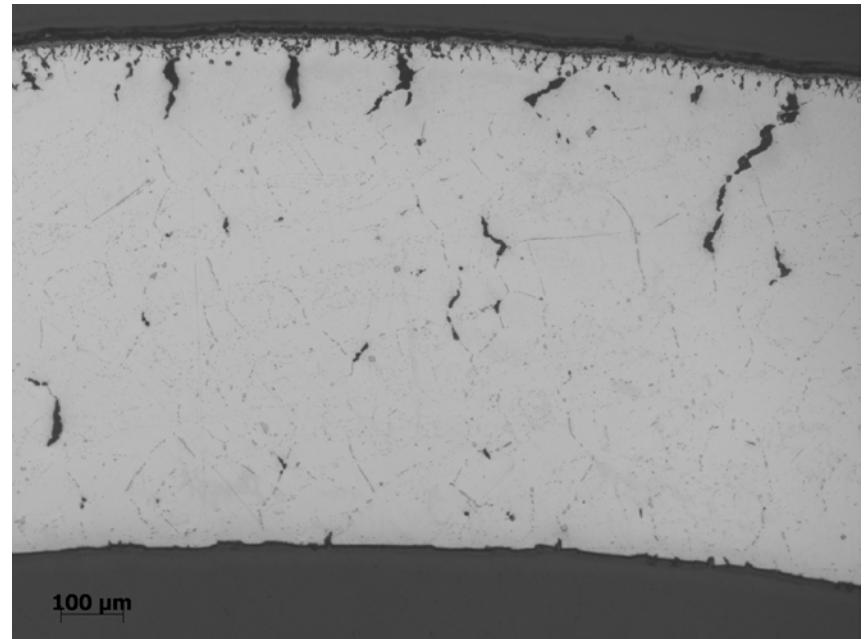
Comparison of Pressurized Tube to Uniaxial Creep



Optical Micrographs of Crept Tube

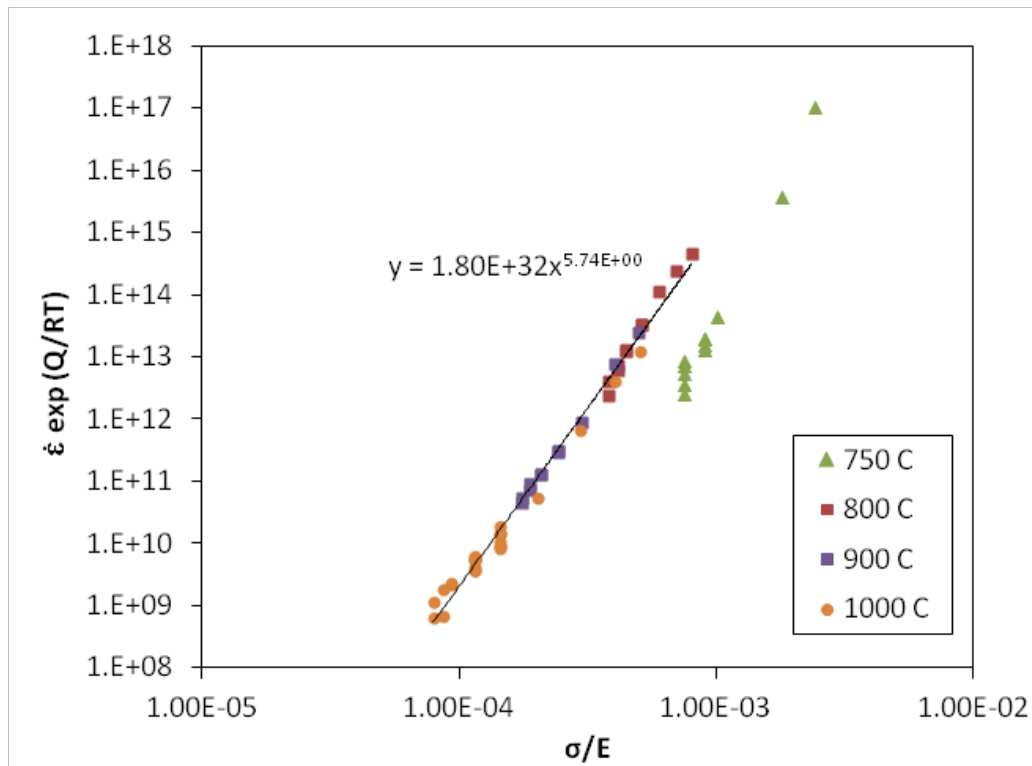


6% strain



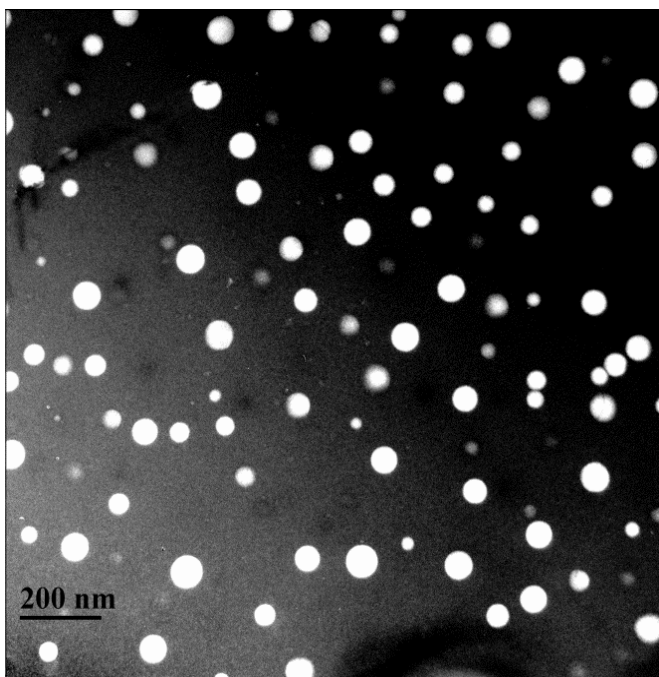
8% strain

Zener-Holloman Plot for Creep of Alloy 617

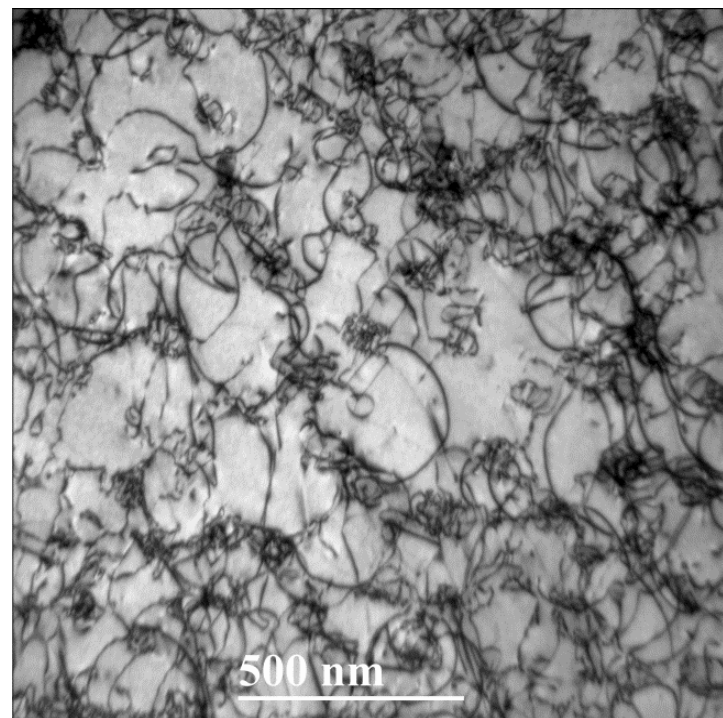


$$\dot{\epsilon} = A\sigma^n e^{-Q/RT}$$

γ' Formation and Interaction in Creep in Alloy 617



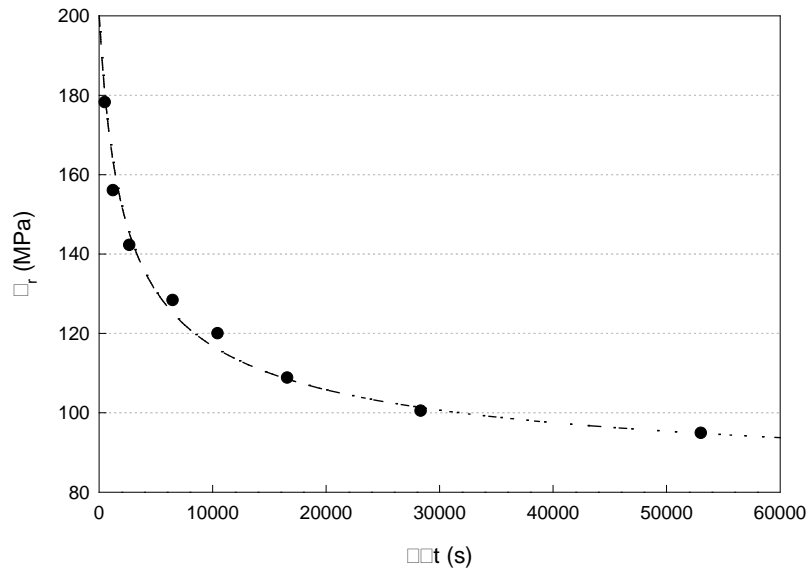
TEM darkfield image



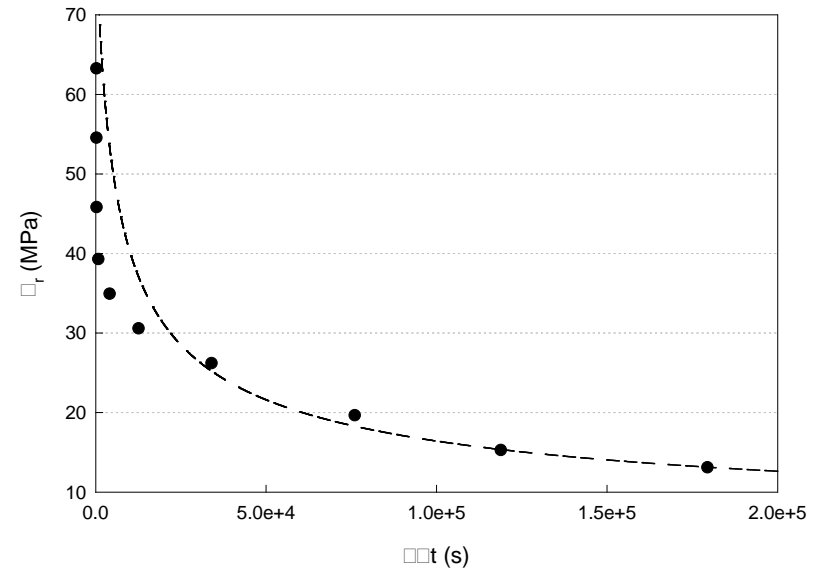
750°C and 145 MPa, ~10% creep strain, 2127 hrs

Uniaxial Creep Stress Drop Tests

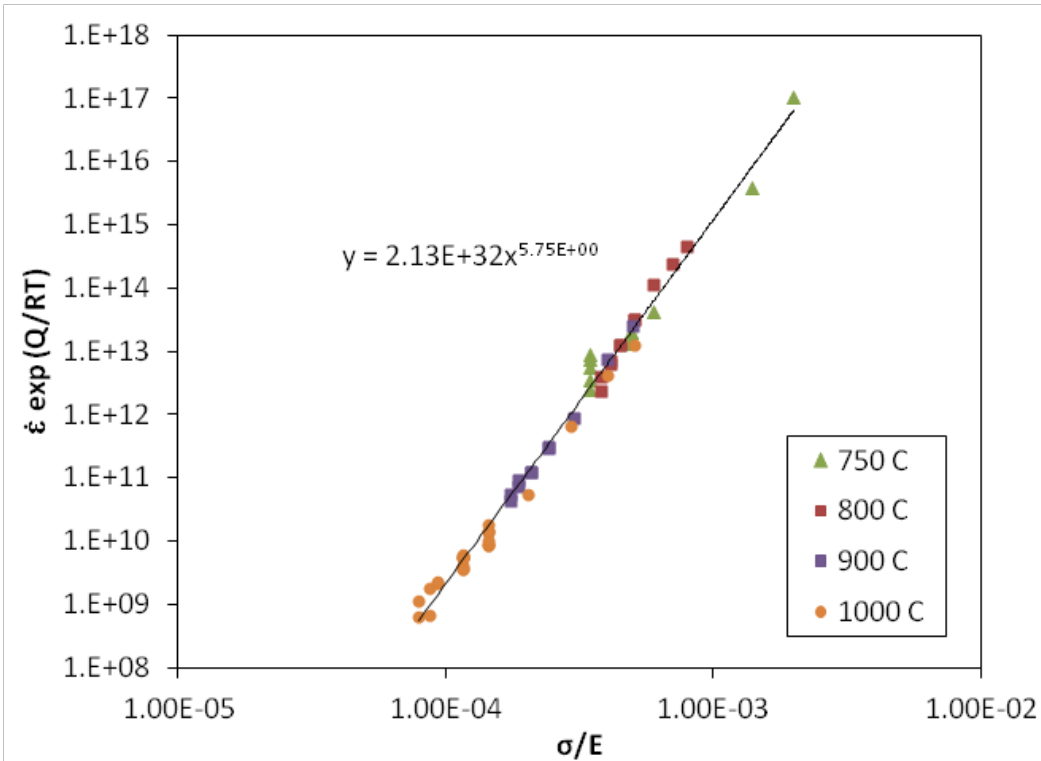
750°C



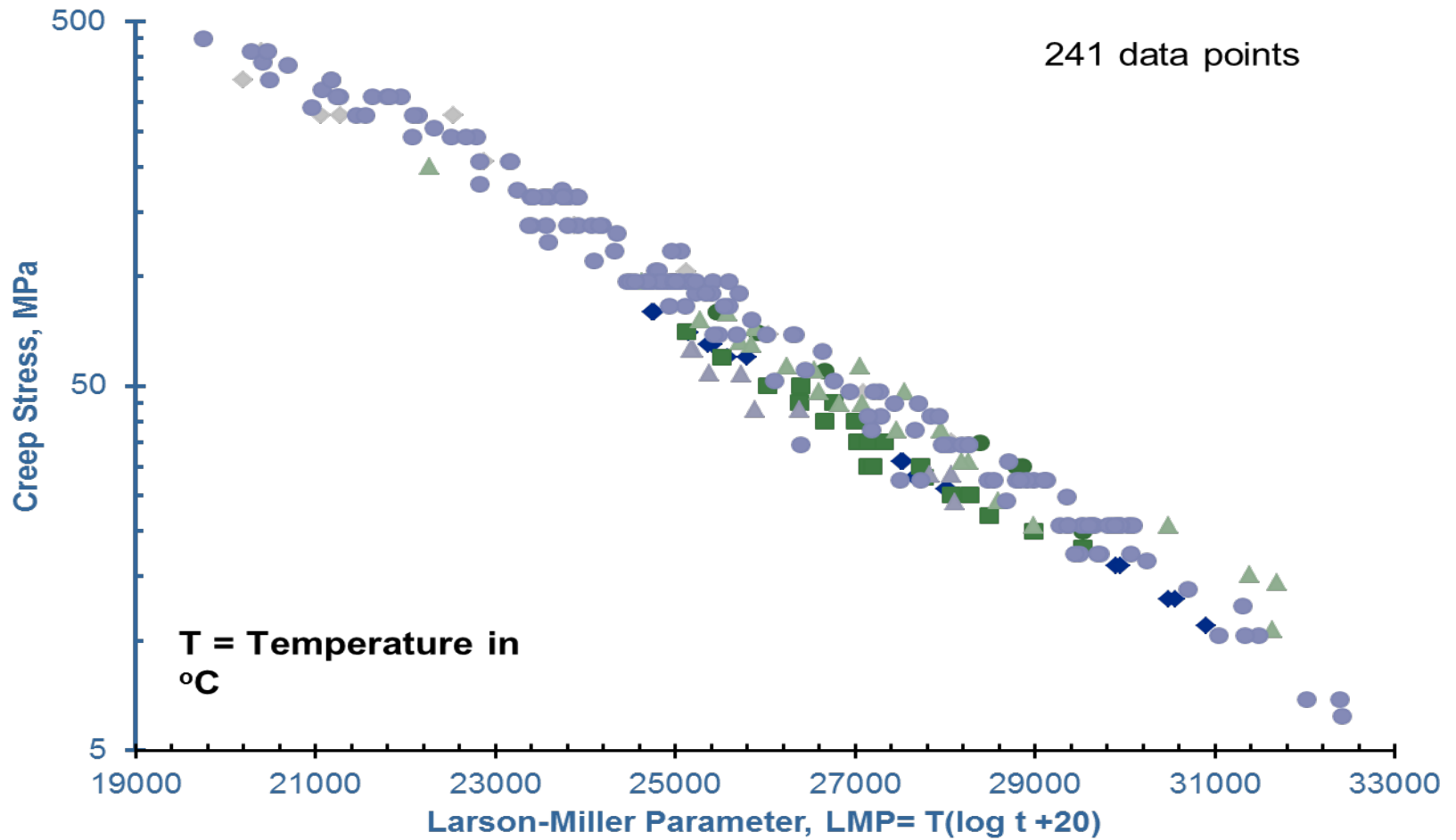
900°C



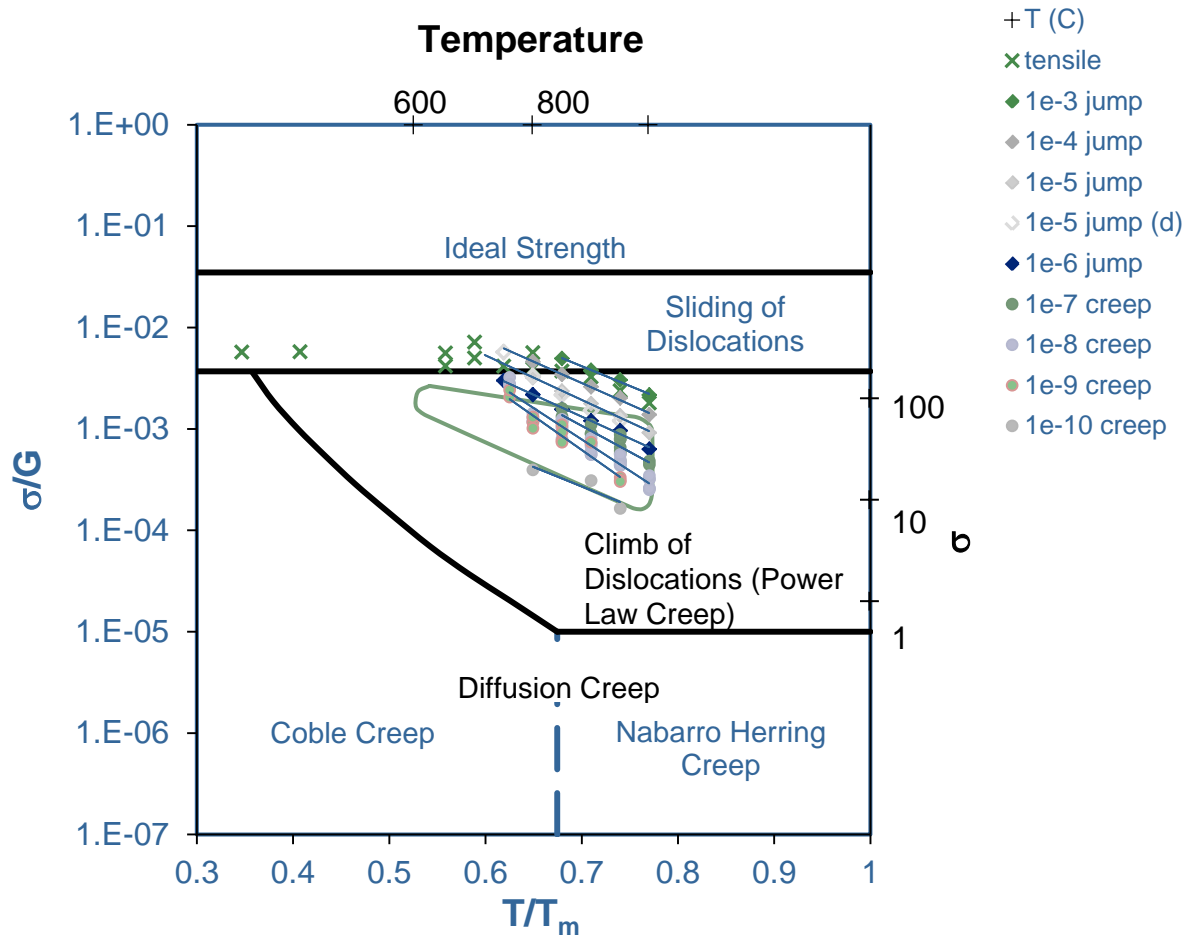
Zener-Hollaman with Threshold Stress at 750°C



$$\dot{\epsilon} = A(\sigma_{\text{applied}} - \sigma_{\text{threshold}})^n e^{-Q/RT}$$

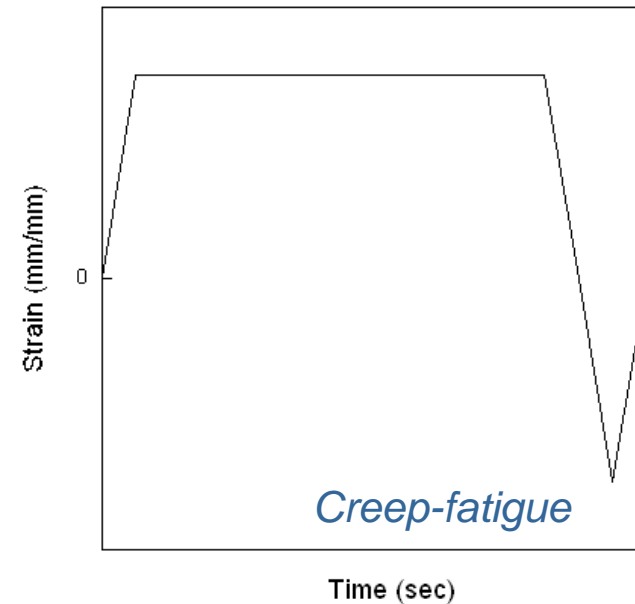
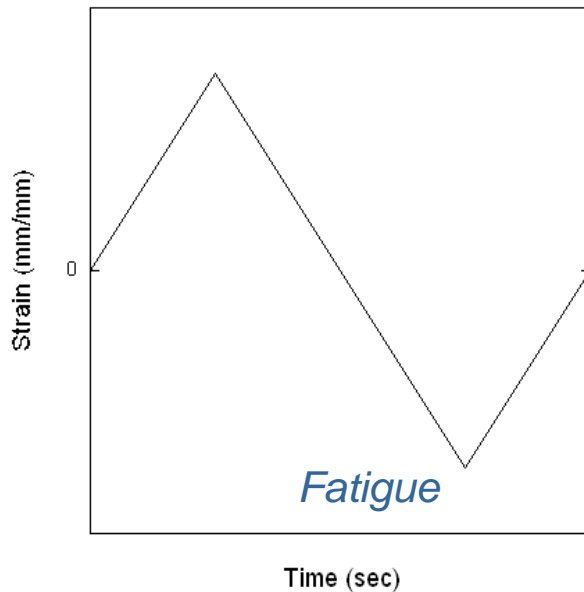


Alloy 617 Deformation Mechanism Map



Heinz-Josef Penkalla, Hans-Helmut Over, and Florian Schubert, "Constitutive Equations for the Description of Creep and Creep Rupture Behavior of Metallic Materials at Temperatures above 800°C" Nuclear Technology, Vol. 66, Sep 1984, p. 685.

- 950°C Creep-Fatigue (CF) behavior
- 850°C C-F behavior
- 950°C C-F and creep behavior of welds



C-F (Tensile-Hold) Failure Modes

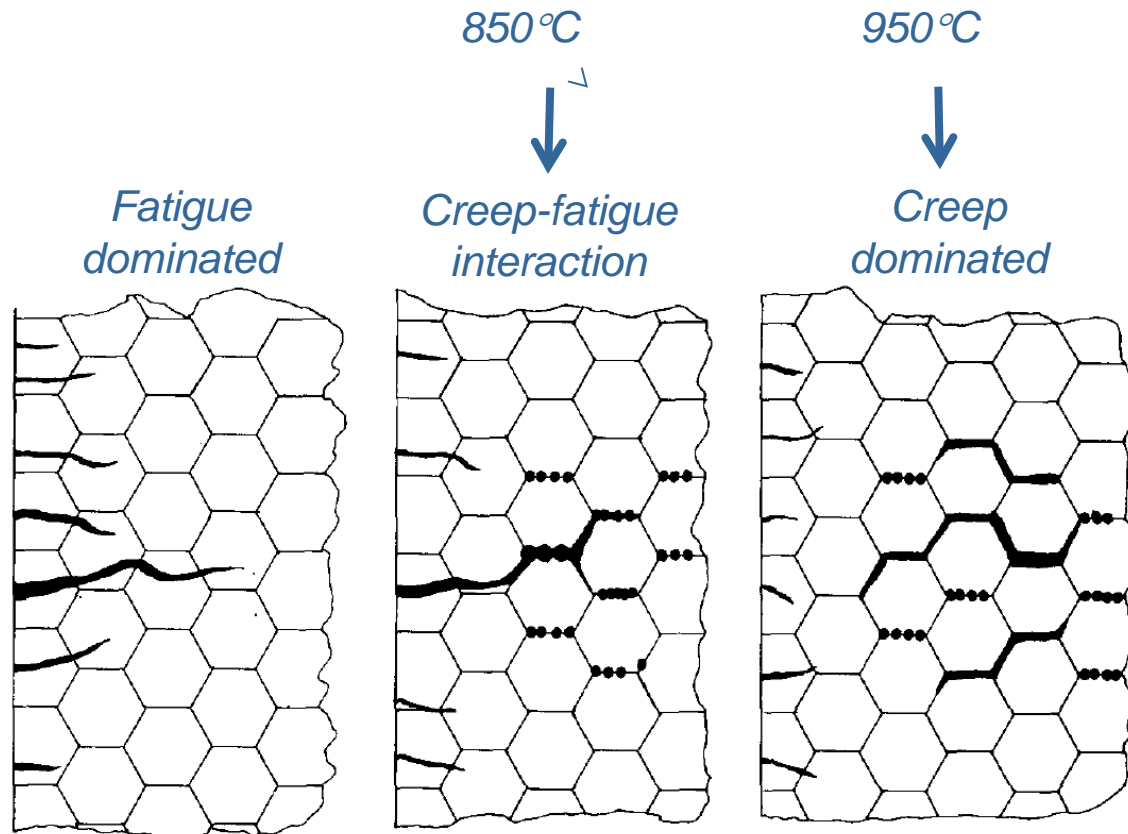
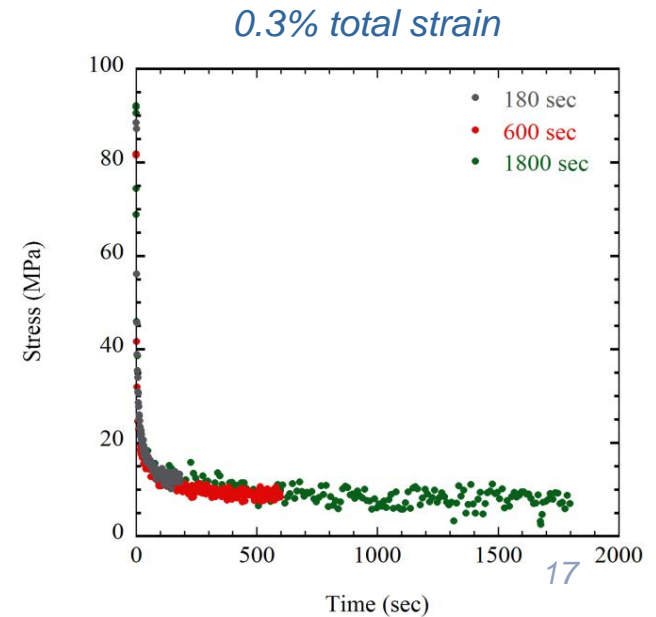
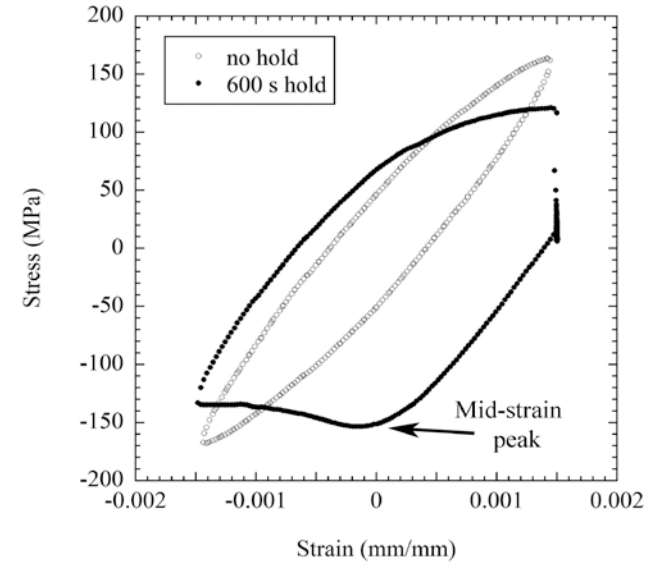
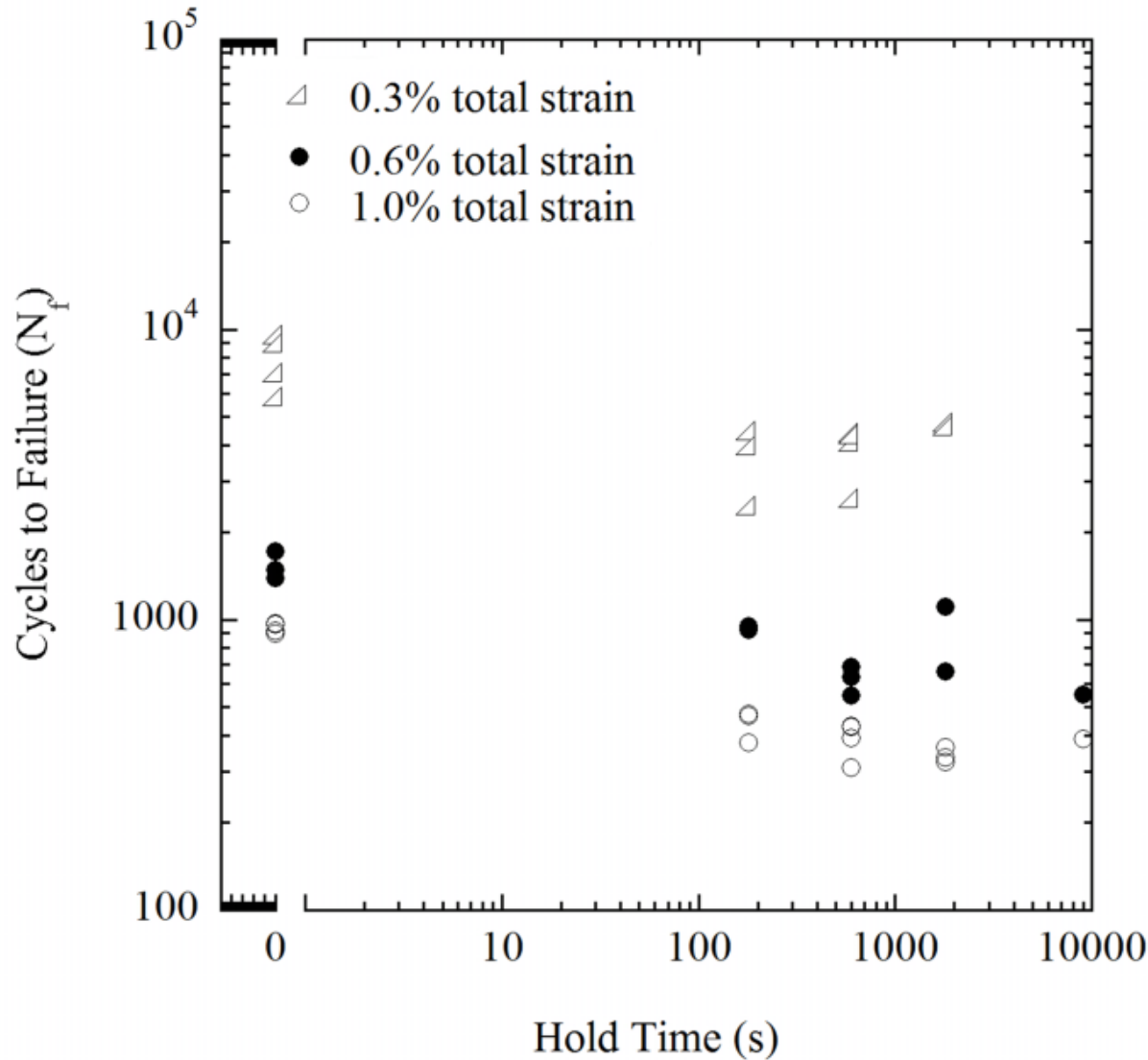
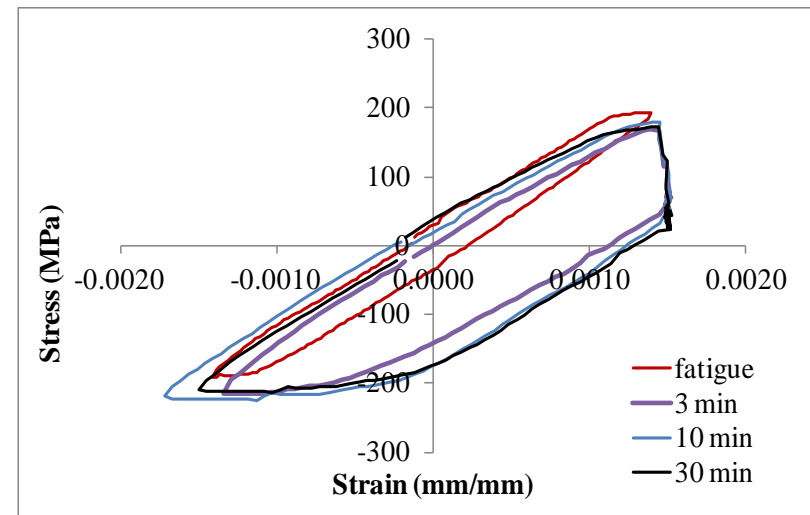
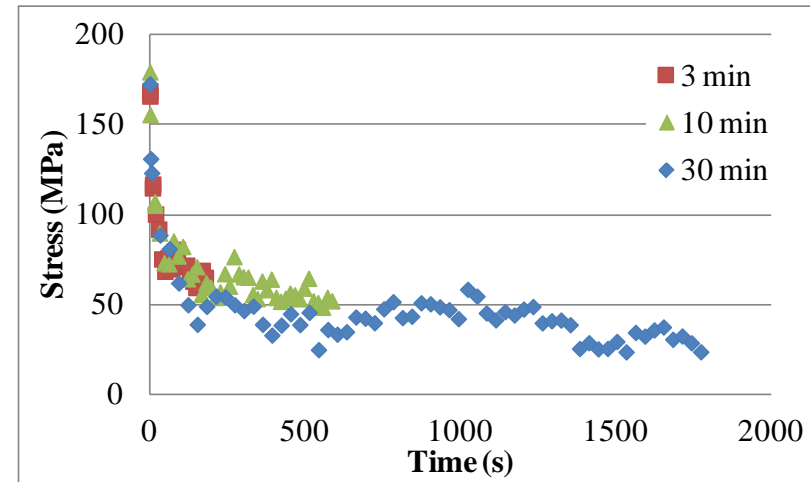
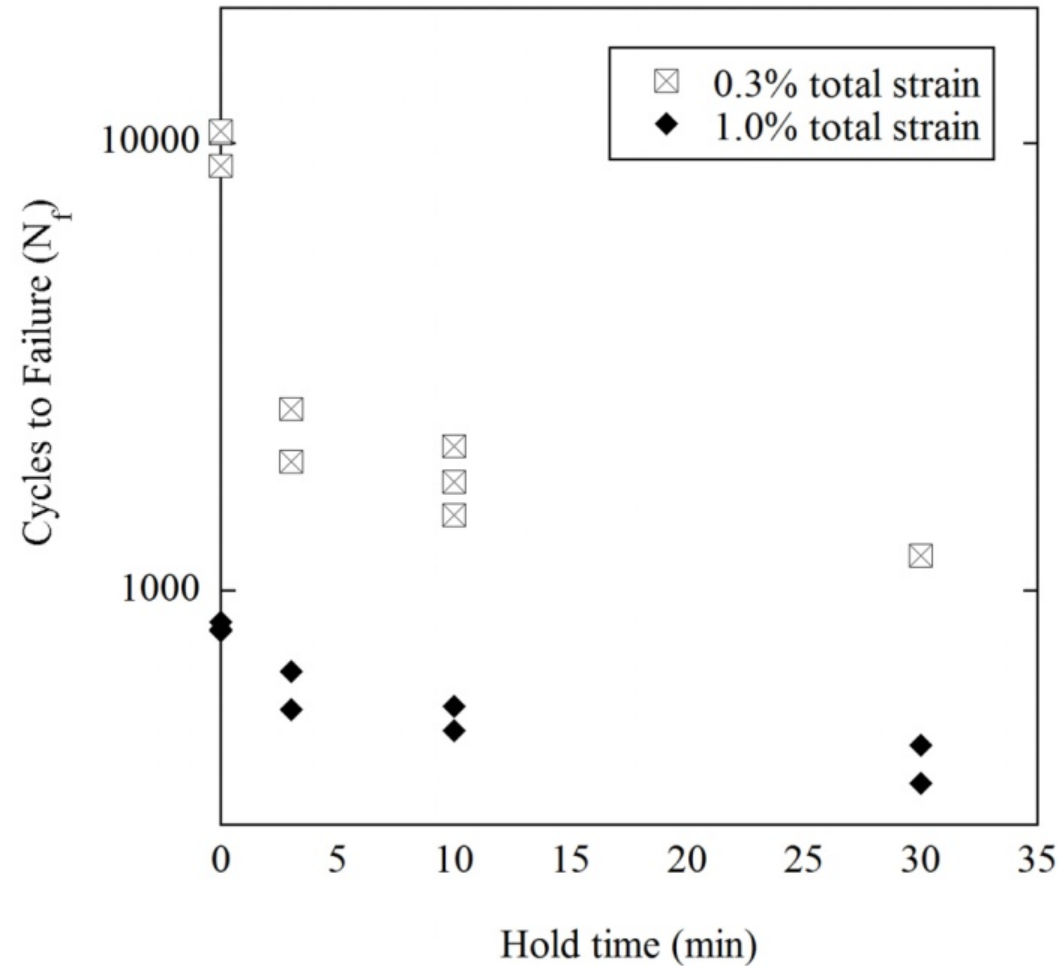
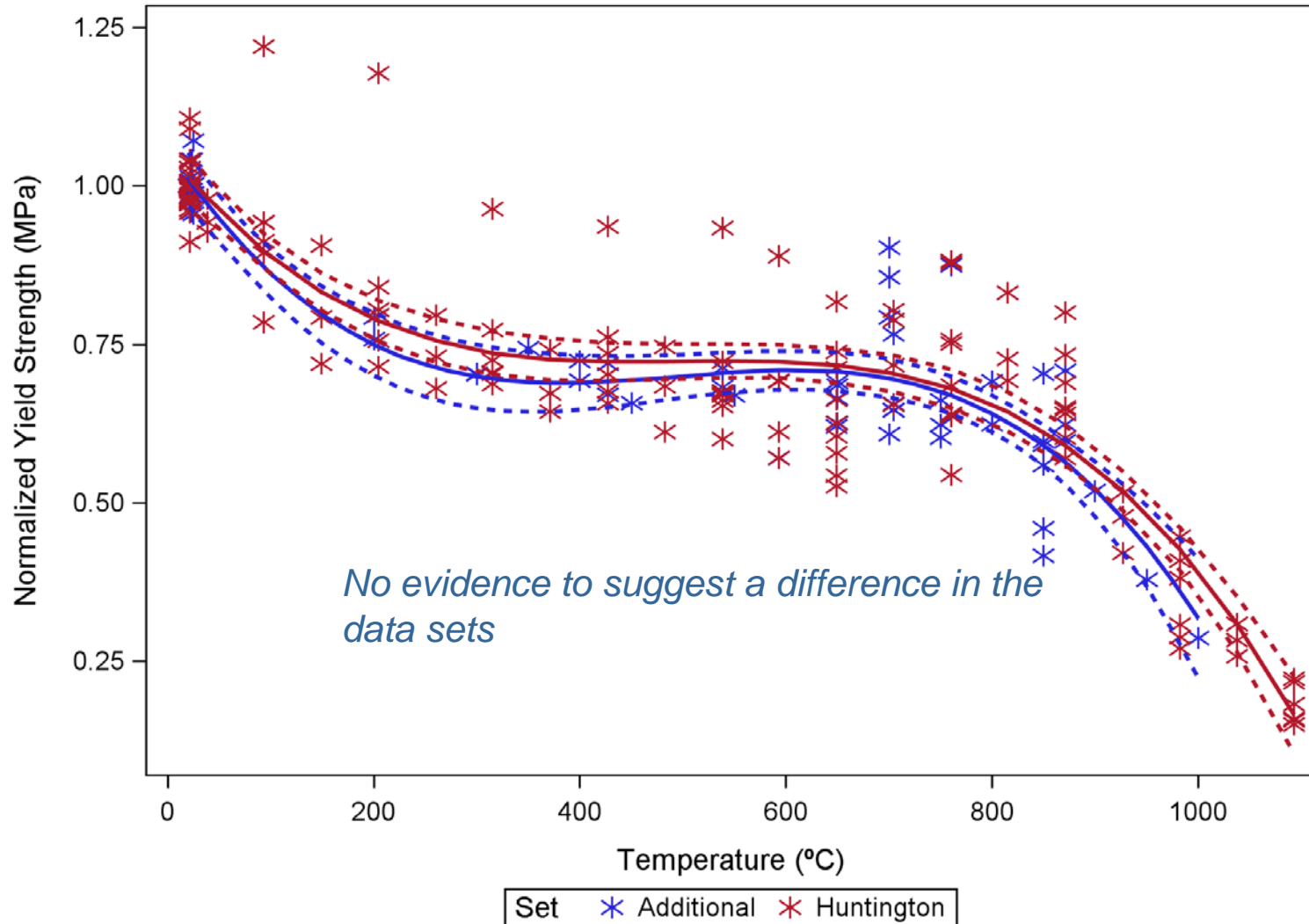


Figure from Miller, Hamm, Phillips, *Materials Science and Engineering*, vol. 53, p.234.

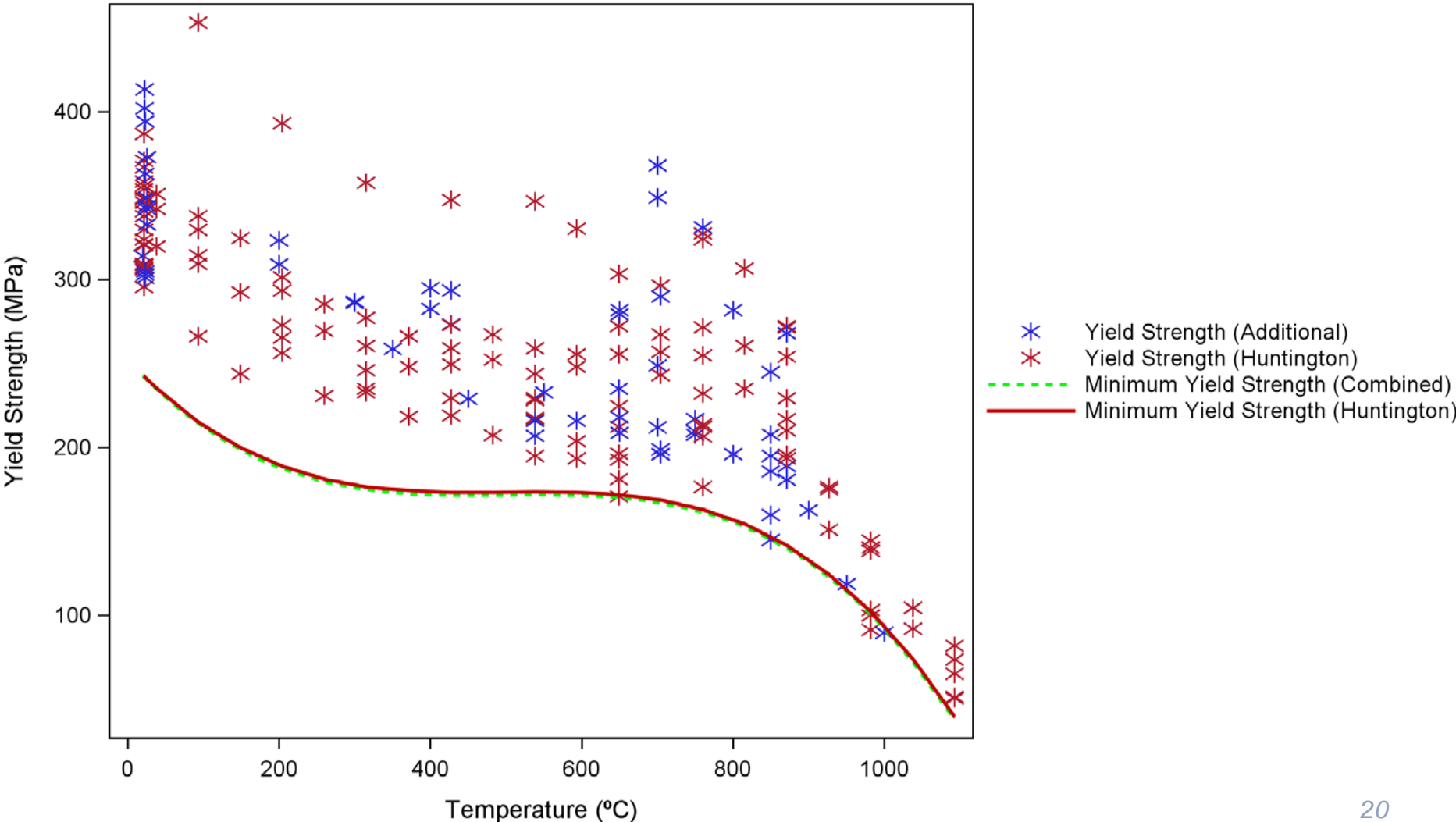




Statistical Equivalence of New Tensile Yield Data

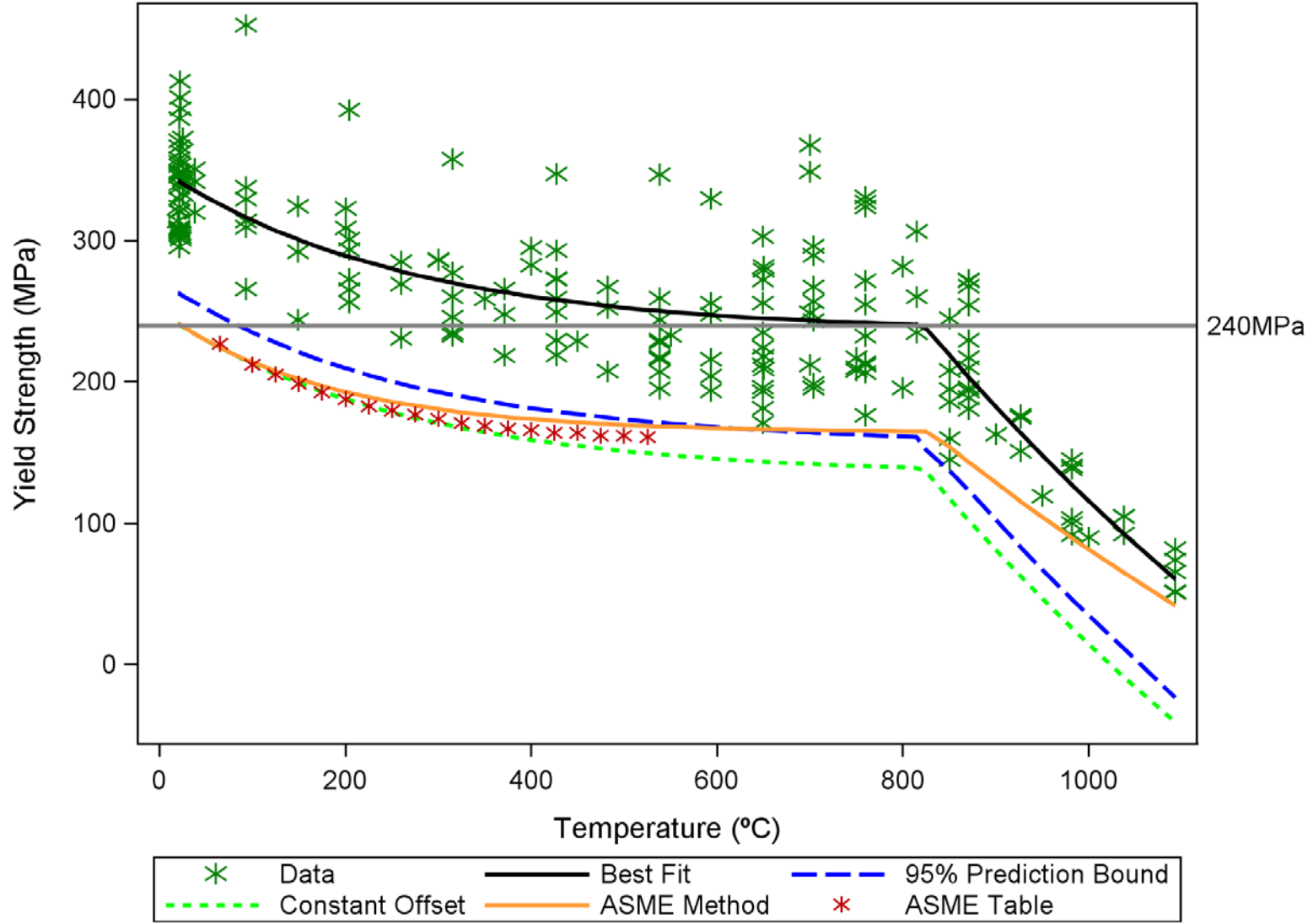


ASME Minimum Yield Strength

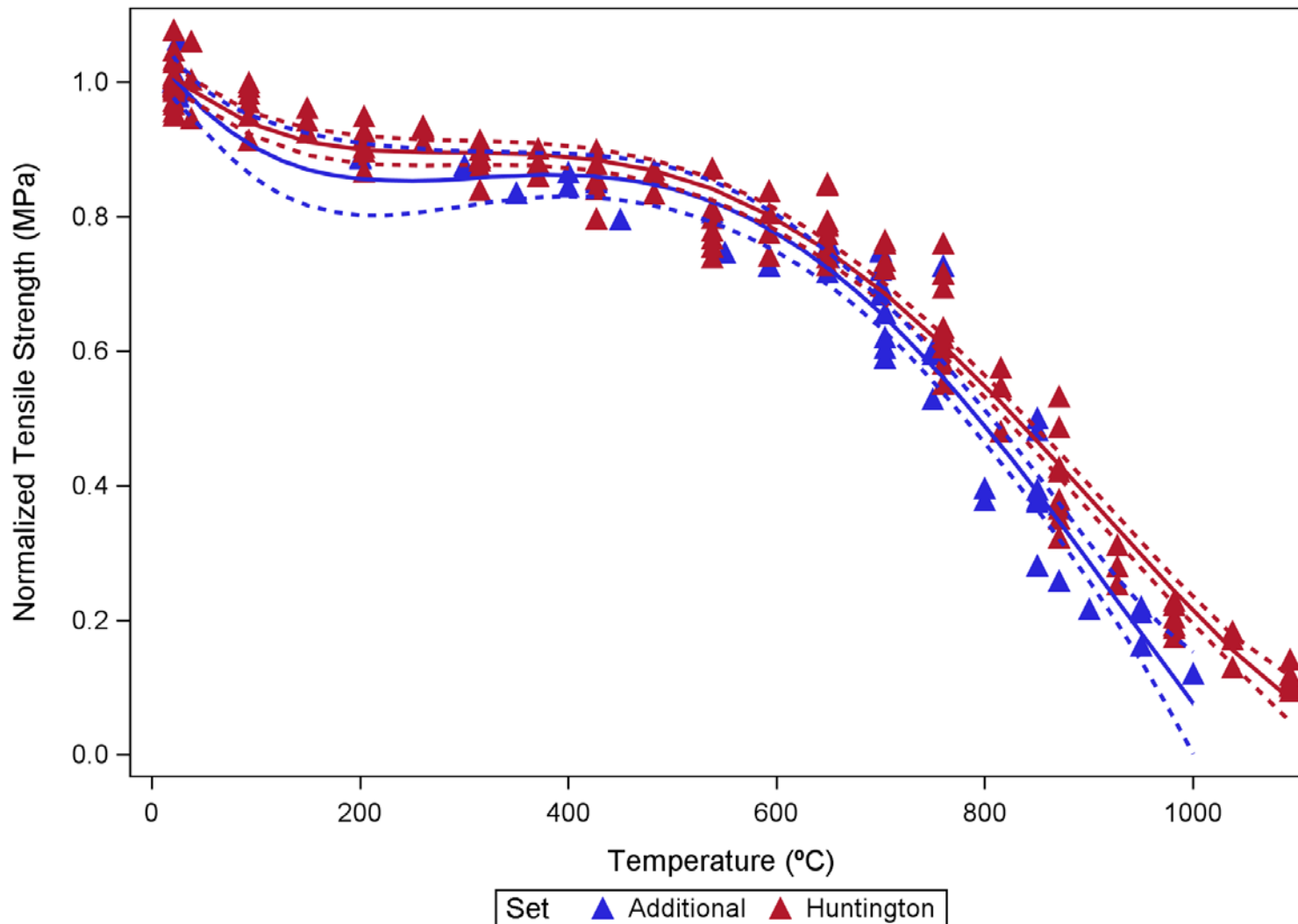


Comparison of Different Methods

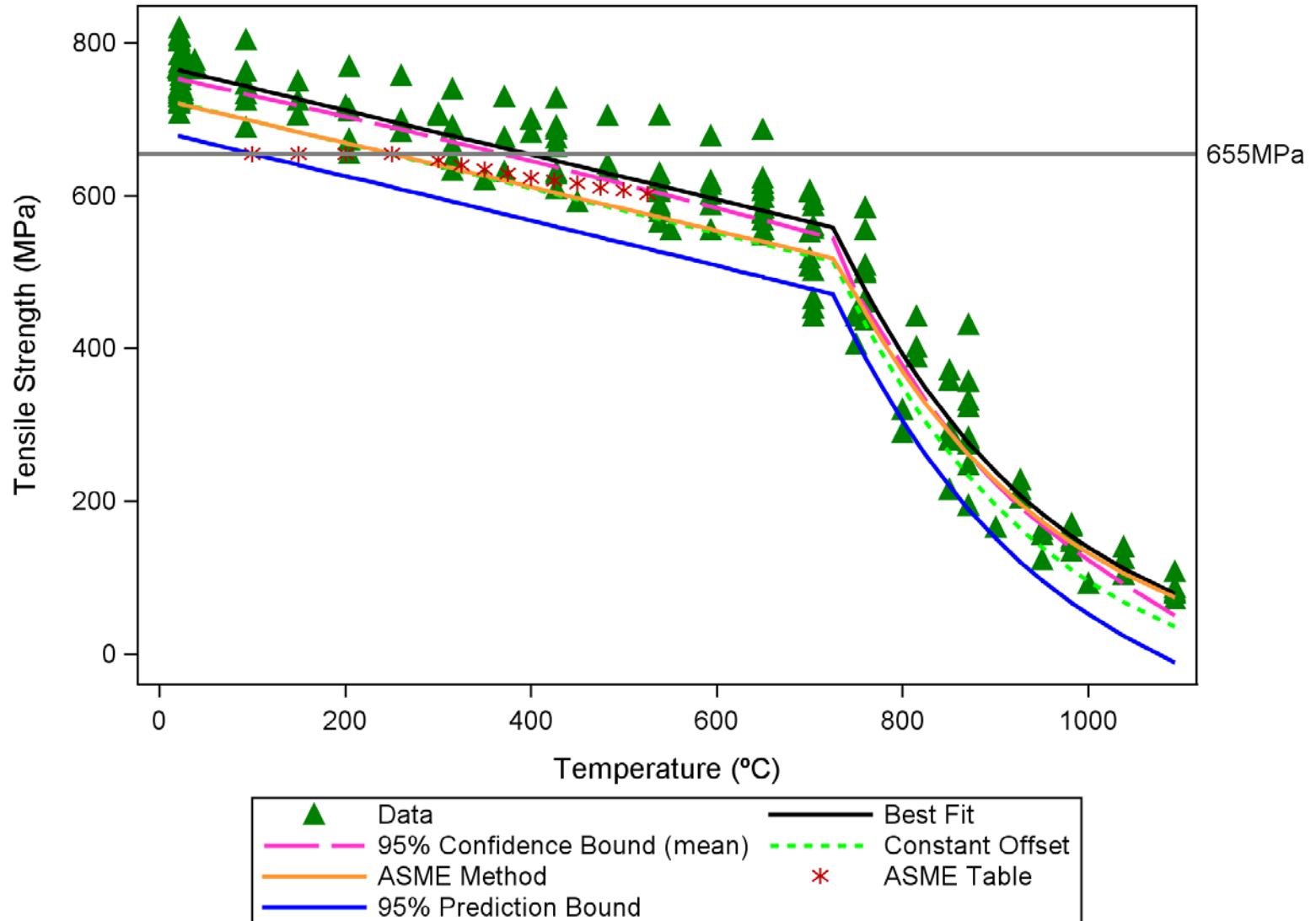
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Statistical Equivalence of New UTS Data

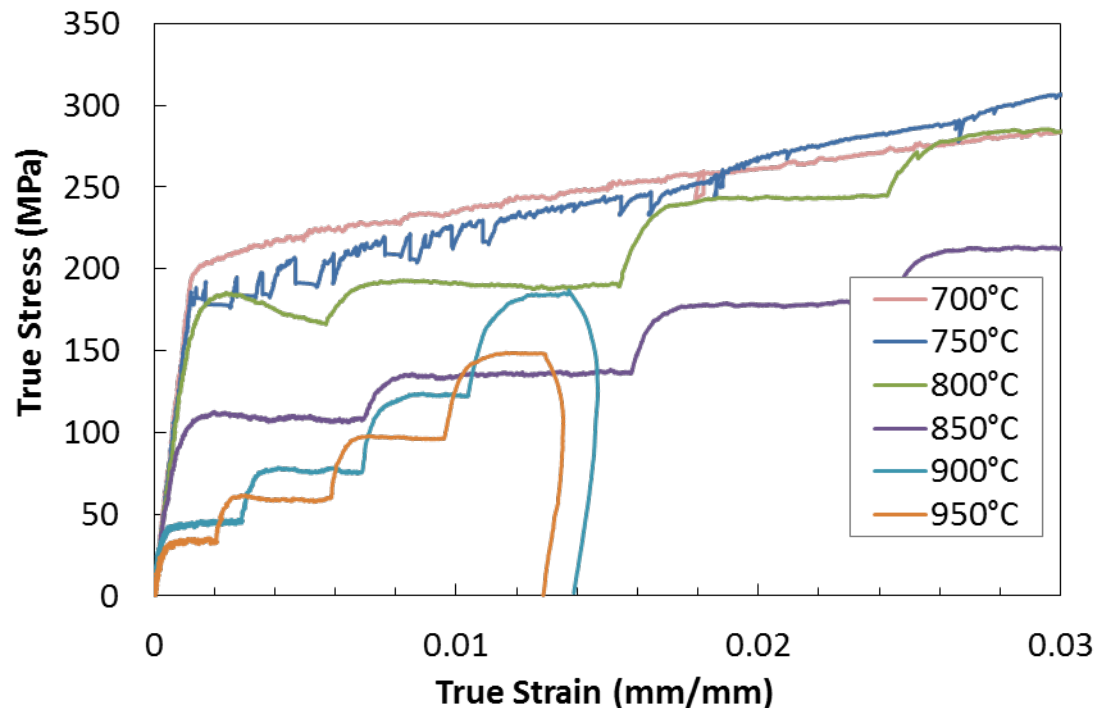


Comparison of Different Methods



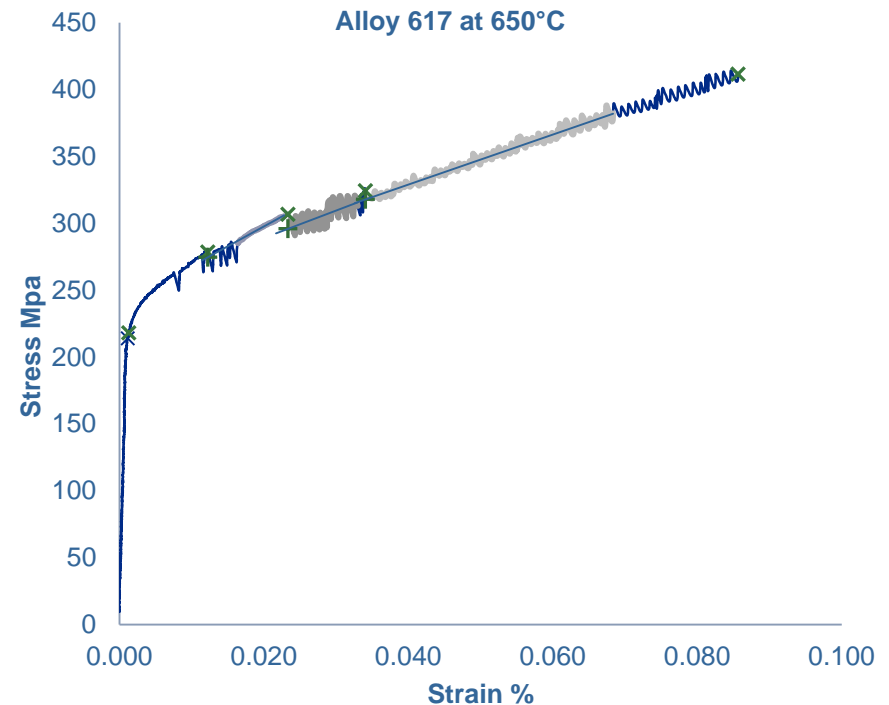
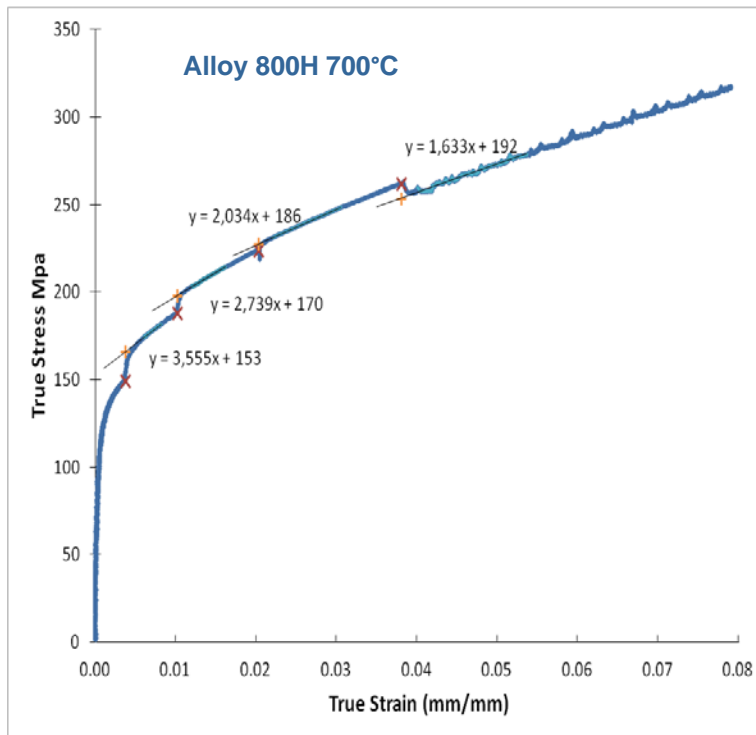
Strain Rate Jump Tests, Alloy 617

- Above 800°C the increments of stress associated with strain rate jumps are of consistent magnitude
- Serrated flow associated with dynamic strain aging occurs at lower temperatures, but little strain rate sensitivity



Negative Strain Rate Sensitivity and Serrated Flow

- Dynamic strain aging (serrated flow) results in negative strain rate sensitivity at lower temperatures
- In this case the material exhibits *strain rate weakening* – the material in the neck is weaker than adjacent material and necking is exacerbated
- Strain rate enhanced necking results in decreased elongation



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- *ASME Code qualification of Alloy 617 is moving forward first for NB and by FY 15 for NH*
- *Preliminary fatigue design curve is under development*
- *It has been demonstrated that γ' can yield creep strengthening at 750°C*
- *Creep-fatigue behavior undergoes a distinct transition between 850 and 950°C*
- *Tensile properties for contemporary material compare well with historical data – although a new method of specifying minimum yield strength may be desirable*