

# Investigations of Interfacial Structure in Thermoelectric Tellurides

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# Collaborators for this work

## **Sandia:**

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## **Caltech:**

Nick Heinz  
Teruyuki Ikeda  
G. Jeffrey Snyder

## **LBNL:**

Quentin Ramasse  
Currently: SuperSTEM, Daresbury, UK

## ***Special thanks to:***

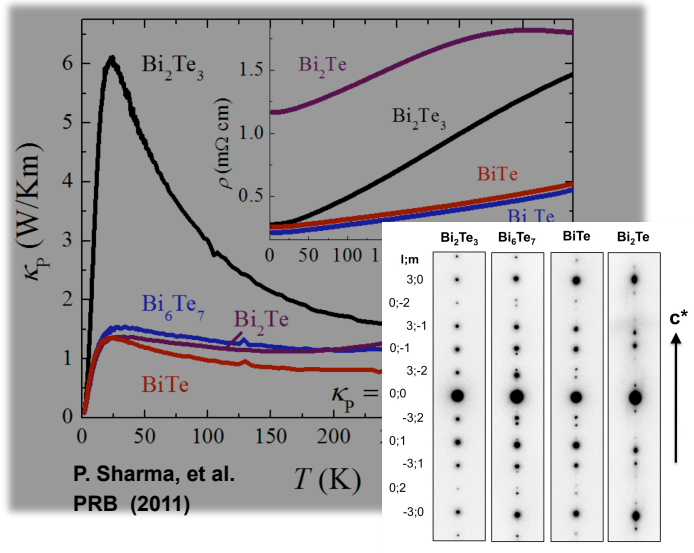
- LBNL: User program, National Center for Electron Microscopy
- LLNL: John Bradley, for use for LLNL's Titan 80/300 instrument
- UCD: Z. Zhang and E. Lavernia, for assistance with SPS processing

Sandia is a multiprogram laboratory operated by Sandia Corporation, a Lockheed-Martin Company, for the United States Department of Energy, National Nuclear Security Administration under Contract DE-AC04-94AL85000.

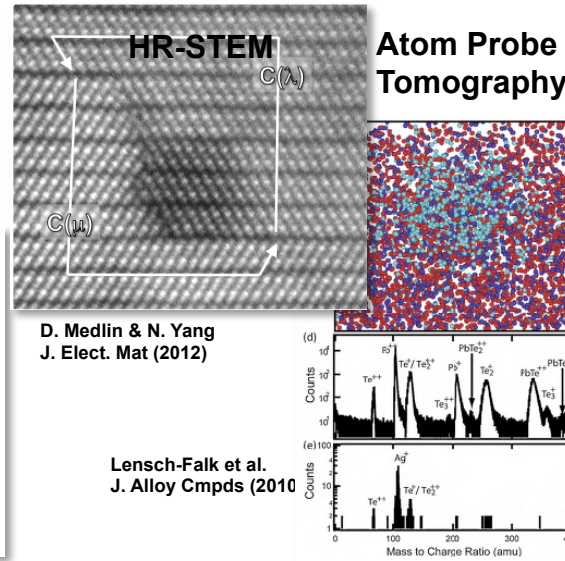


# Thermoelectric Materials Science at Sandia: Fundamental Science to Device Level Detail

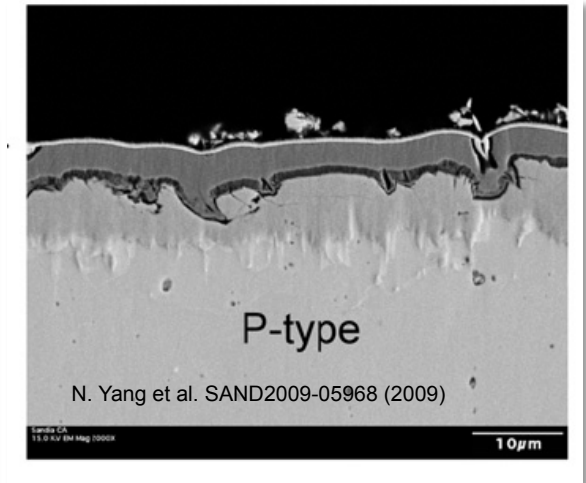
## Transport Physics



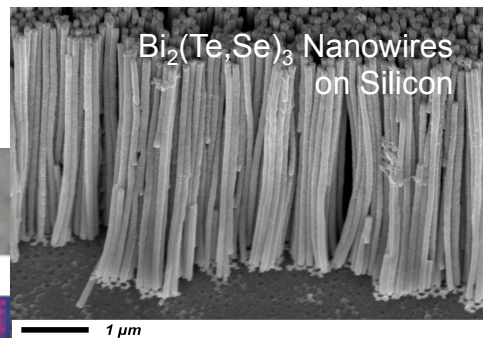
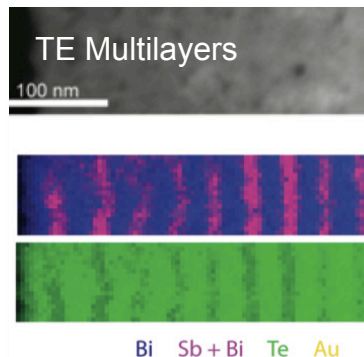
## Advanced Microscopy



## Aging & Compatibility



## Materials Synthesis

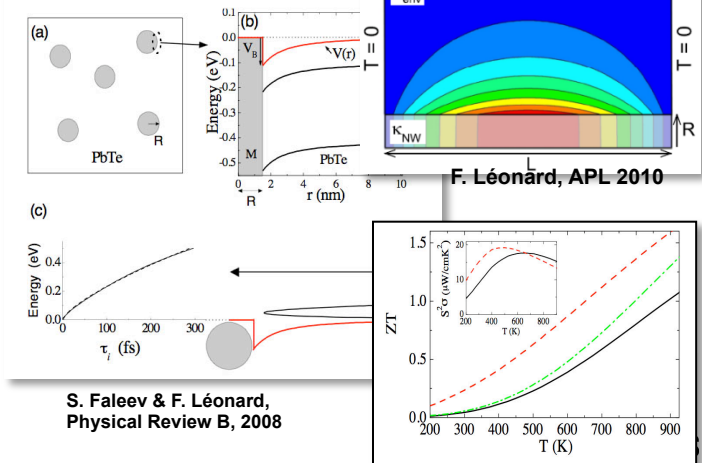


S. J. Limmer, et al. J. Electrochem Society. (2012).

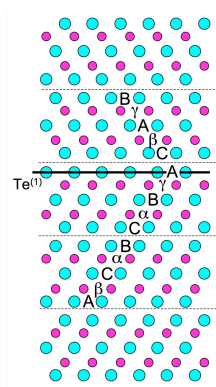
D. Banga et al. J. Cryst. Growth & Des. (2012).

## Theory and Modeling

### Transport



### Structure



D.L. Medlin et al, J.Appl. Phys (2010)

# Important Roles for Interfaces in Thermoelectric Materials

## Interfacial strategies to enhance ZT:

-phonon scattering:  $\downarrow K$

-energy filtering:  $\uparrow S^2\sigma$

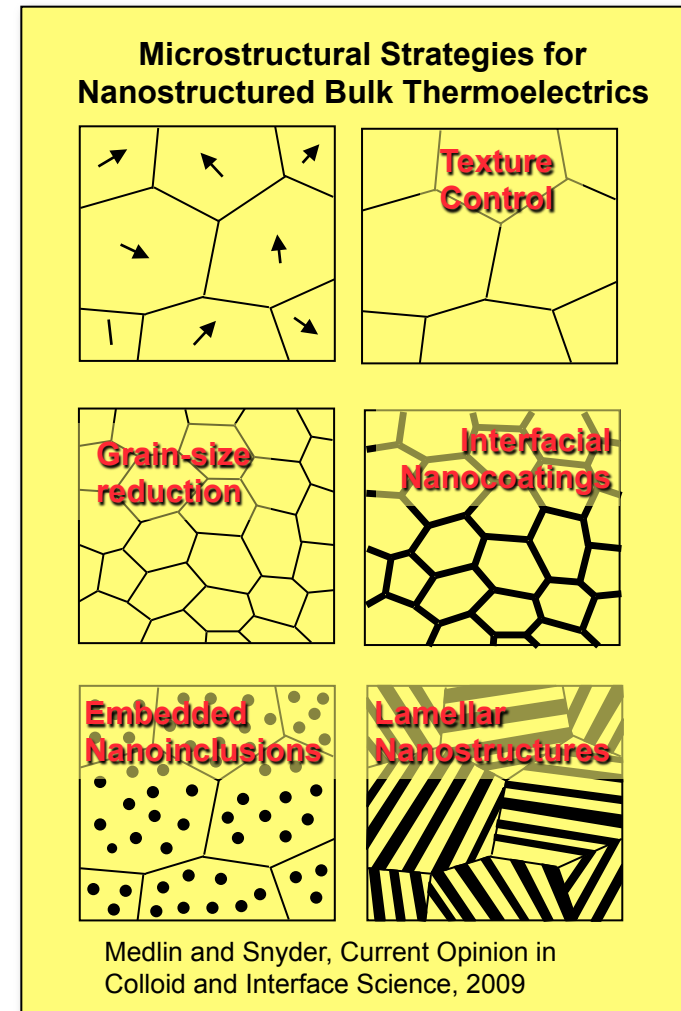
## Bulk Thermoelectrics:

- grain size reduction
- solid state phase transformations

*What are mechanisms of interface formation and stability?*

*How do interfaces control local transport properties?*

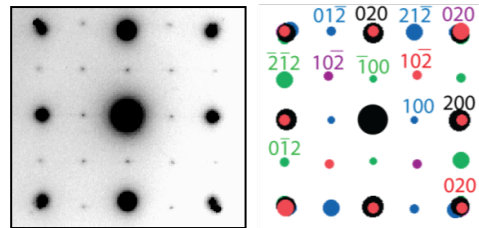
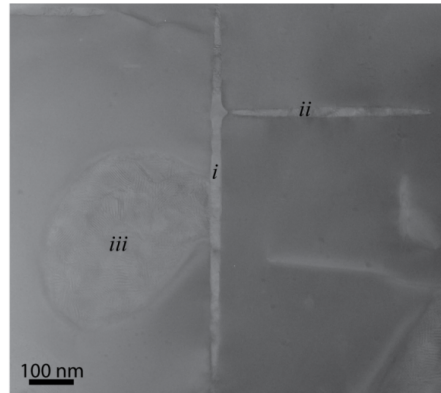
***Needed: foundational understanding of interfacial structure and composition in thermoelectric materials***



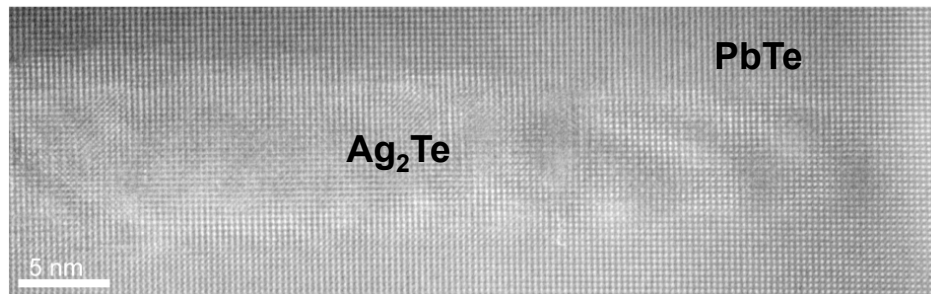


# Example: Enhanced zT in Ag<sub>2</sub>Te/PbTe system

Crystallographically aligned precipitates of Ag<sub>2</sub>Te in thermoelectric PbTe matrix



[201]monoclinic || [001]fcc  
(-204)monoclinic || (2-20)fcc  
+ symmetry-related variants



J. Lensch-Falk, J. Sugar, M. Hekmaty, D. Medlin,  
Journal of Alloys and Compounds (2010)

Initial precipitates form topotactically at high T in FCC phase

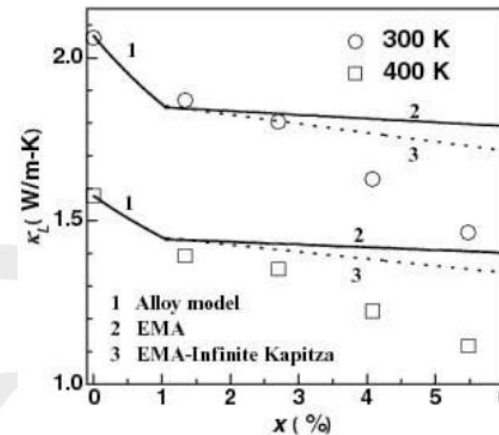
Transform to monoclinic on cooling to room temperature

Plate morphology relieves elastic strain.

Same orientation relationship observed for Ag<sub>2</sub>Te in AgSbTe<sub>2</sub>.

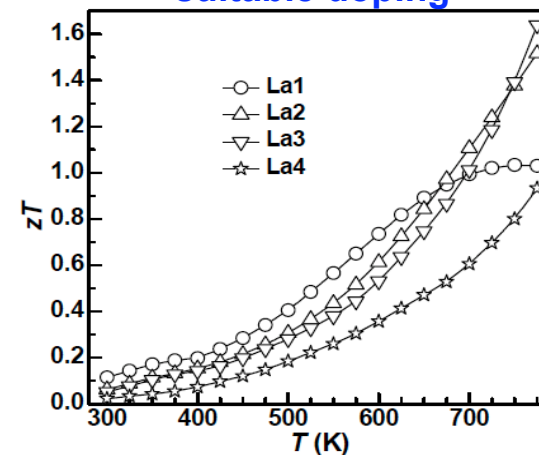
J. Sugar & D. Medlin, Journal of Alloys and Compounds (2009)

Ag<sub>2</sub>Te precipitates reduce thermal conductivity by phonon scattering



Collaboration  
with Snyder  
Group, Caltech

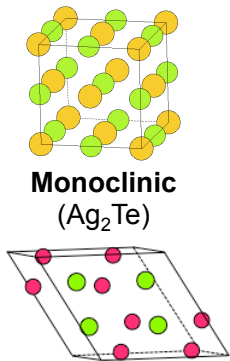
zT as high as 1.6 with  
suitable doping



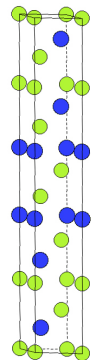
Y. Pei, J. Lensch-Falk, E.S. Toberer, D.L. Medlin,  
G.J. Snyder, Advanced Functional Materials (2011).

# Focus for this talk: Interfacial structure in Thermoelectric Tellurides

**Rock-Salt**  
(PbTe, AgSbTe<sub>2</sub>)



**Tetradymite**  
(Bi<sub>2</sub>Te<sub>3</sub>, Sb<sub>2</sub>Te<sub>3</sub>)



*Rich set of structures and phase relations*

*Can we begin to make sense of interfaces in these complex systems?*

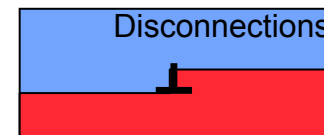
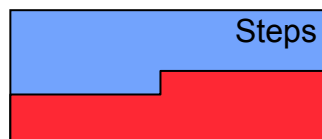
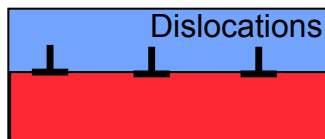
- **Structure of the basal twin in Bi<sub>2</sub>Te<sub>3</sub>**

- comparison with ab initio calculations
- Analysis of a twin boundary defect
- Analogies to twins in FCC materials

- **Rocksalt/Tetradymite Telluride Interfaces**

- AgSbTe<sub>2</sub>/Sb<sub>2</sub>Te<sub>3</sub>
- PbTe/Sb<sub>2</sub>Te<sub>3</sub>
- Tetradymite plate formation.
- Interfacial strain accommodation.

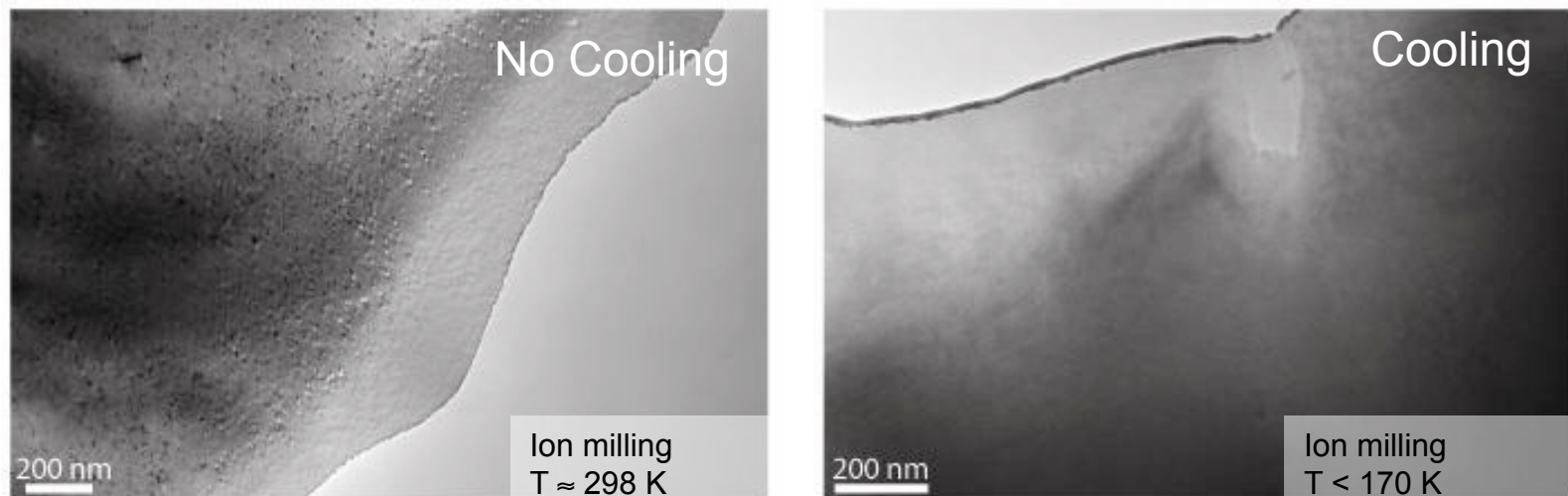
**Interfacial Line Defects:** Building blocks to general understanding of interface structure and behavior.



# An Experimental Challenge: TEM preparation of Tellurides

**Example:** PbTe: Aggressive ion milling conditions produce nanoscale contrast features

PbTe – 99.998% Sigma Aldrich



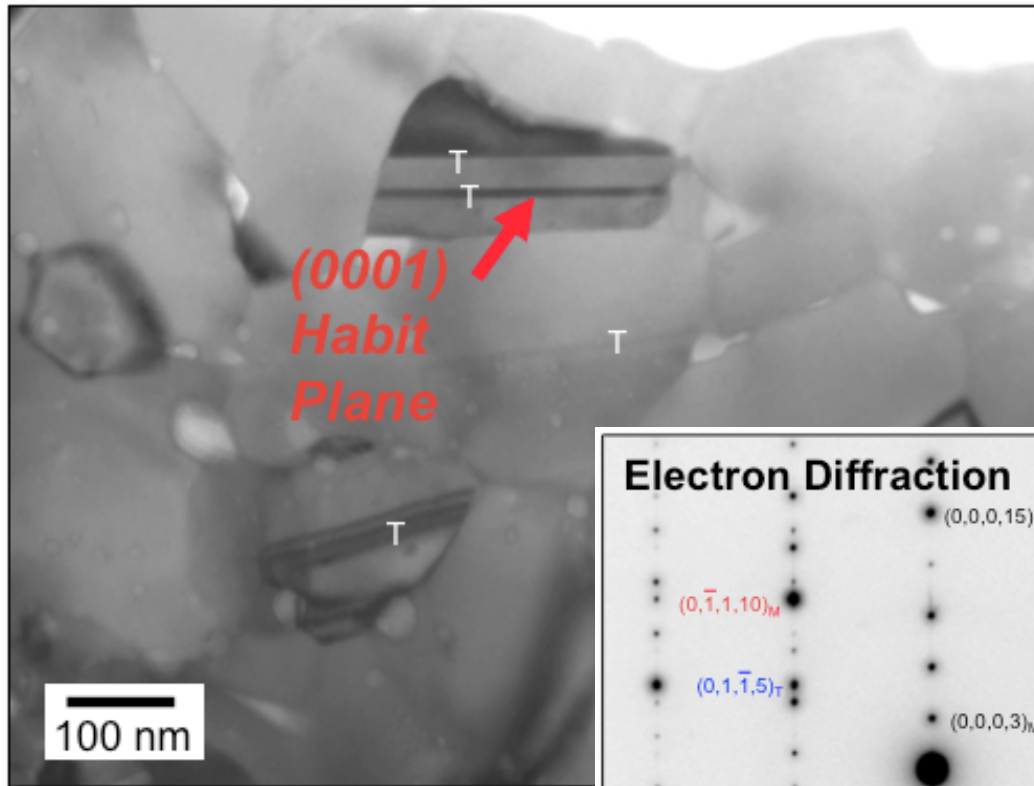
***Artifacts can arise even with cooling if ion-milling power is too high***

**Best conditions are shallow angle (  $\sim 6^\circ$  ) and low power (  $< 1.5$  kV, 3 mA ).**

Fischione 1010 ion mill

J.L. Lensch-Falk, J.D. Sugar, M.A. Hekmaty, D.L. Medlin,  
Journal of Alloys and Compounds, 504 (2010) 37-44

# Basal Twin Boundaries in $\text{Bi}_2\text{Te}_3$

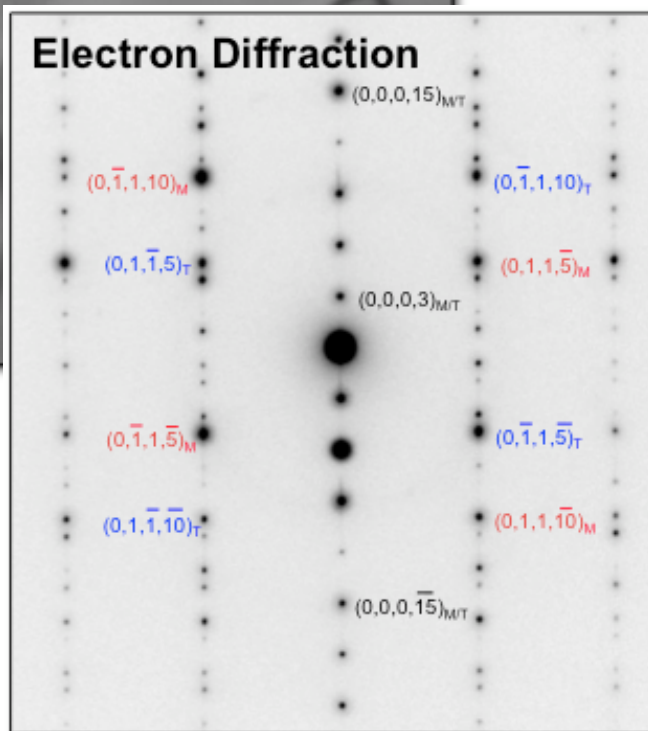


$\text{Bi}_2\text{Te}_3$ : Powder consolidated by spark plasma sintering

TEM Specimen Preparation:  
Low voltage ion milling (1kV)  
Cryo-cooling:  $<-100^\circ\text{C}$

Good starting point for more complex grain boundaries

Potentially favorable TE electronic transport properties  
*-Coherent structure, near bulk-like coordination.*

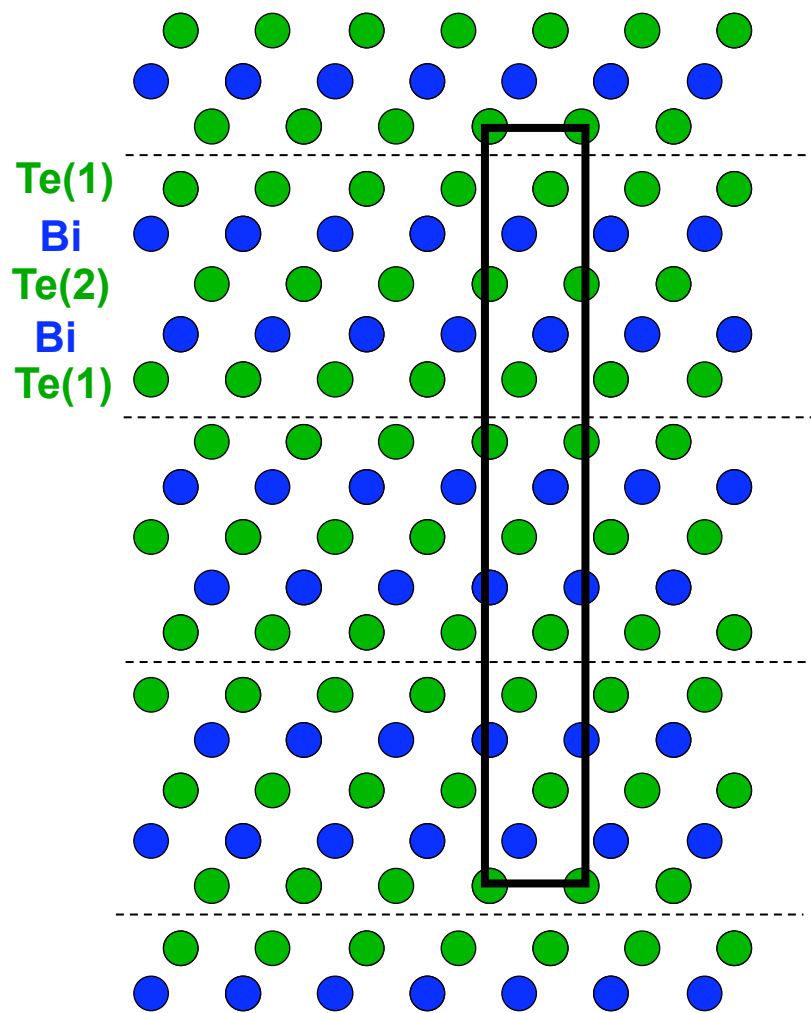


Orientation Relationship:  
 $(0001)//(0001)$   
 $[2-1-10]//[2110]$

***180° rotation about c-axis***

# $\text{Bi}_2\text{Te}_3$ : Three crystallographically distinct basal plane terminations

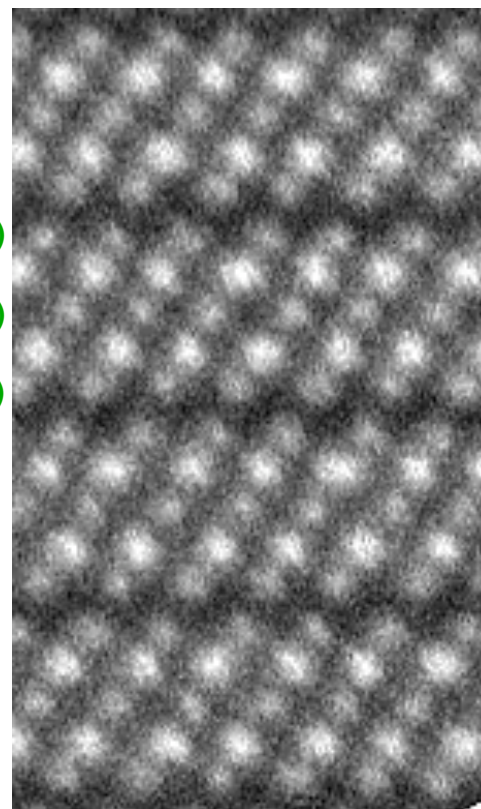
$\langle 2\bar{1}\bar{1}0 \rangle$  projection



- Rhombohedral ( $R\bar{3}m$ ) structure
- Based on tetradymite ( $\text{Bi}_2\text{STe}_2$ ) prototype

HAADF-STEM

Te(1)  
Bi  
Te(2)  
Bi  
Te(1)



Bi:  $Z=83$   
Te:  $Z=52$

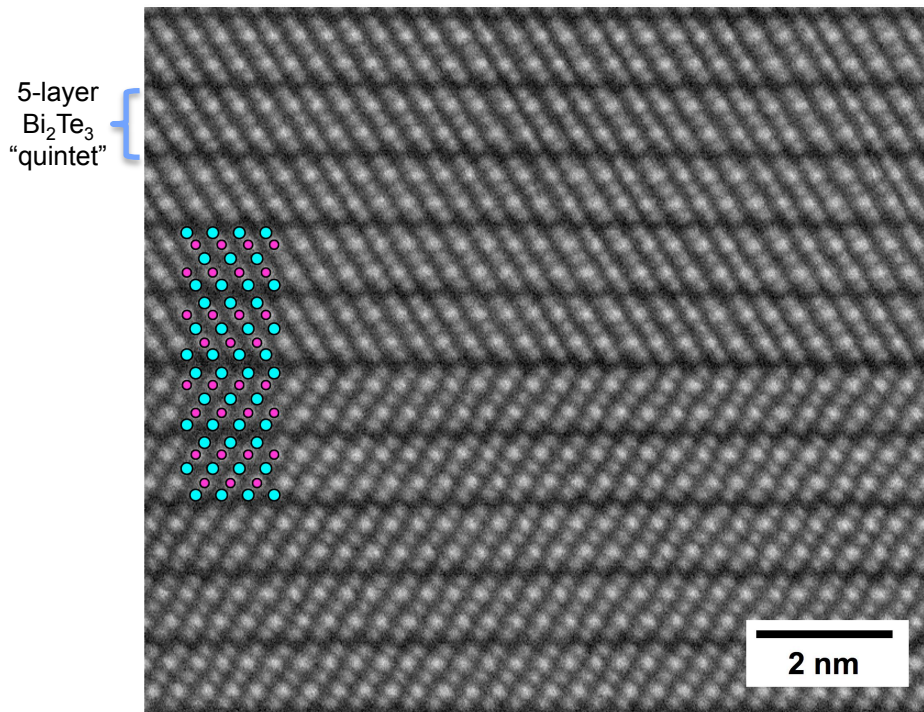
Atomic number difference enables Bi and Te to be distinguished in HAADF-STEM



# Basal (0001) Twin in Bismuth Telluride

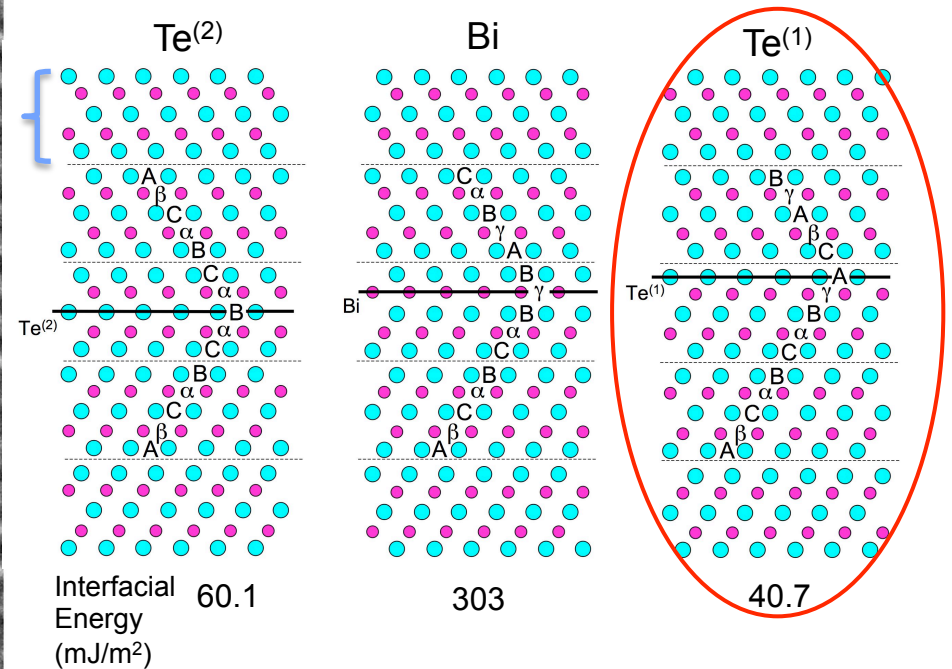
## *Twin boundary terminated at $\text{Te}^{(1)}$ layer*

HAADF-STEM Imaging:



DFT Calculations:

*Three Possible Compositional Terminations*

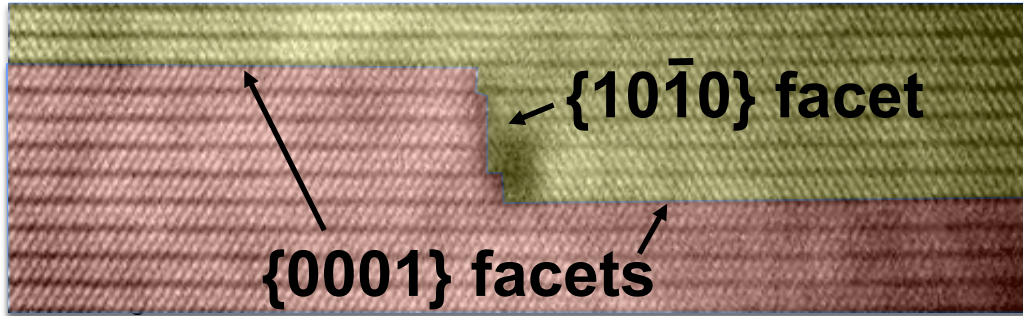


- Strong energetic preference for twin termination at  $\text{Te}^{(1)}$  layer.
- High energy of Bi-terminated interface has implications for twinning mechanism.

D.L. Medlin, Q.M. Ramasse, C. D. Spataru, N.C. Yang, J. Appl. Phys. (2010)

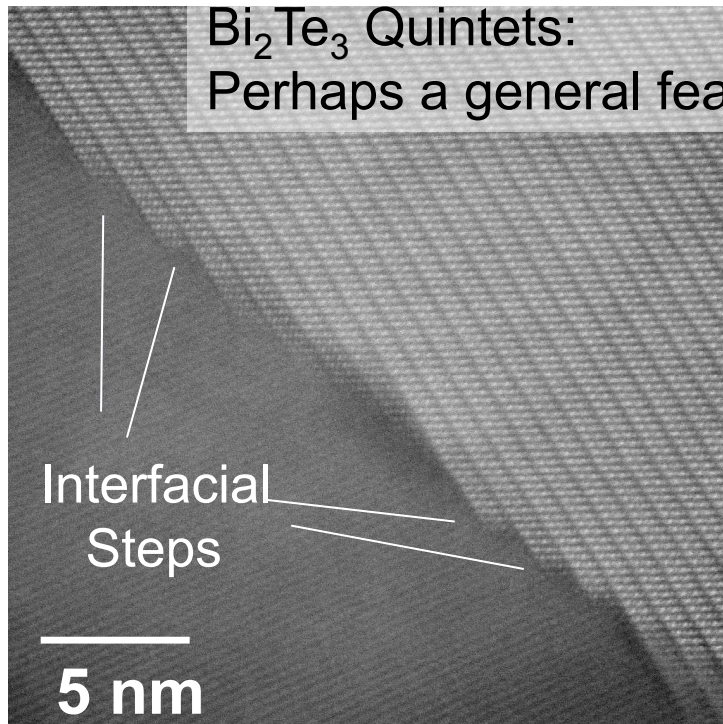


# Steps in the $\text{Bi}_2\text{Te}_3$ Twin Boundary



5 nm

Steps of integral 5-plane  $\text{Bi}_2\text{Te}_3$  Quintets:  
Perhaps a general feature?

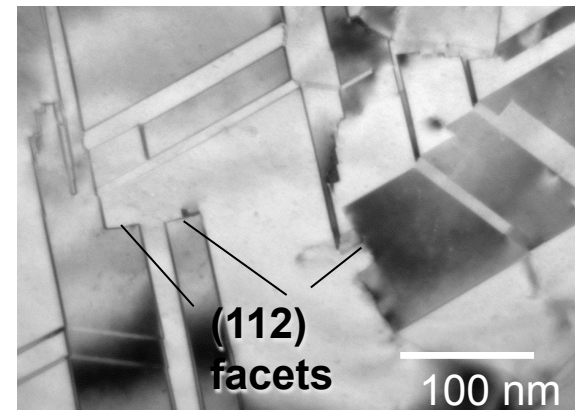


Interfacial  
Steps

5 nm

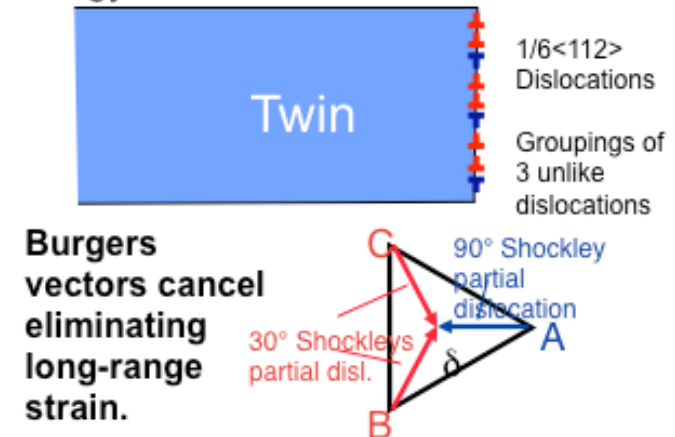
**Morphology analogous to annealing and growth twins in FCC materials.**

Example: Twins in Electrodeposited Ni

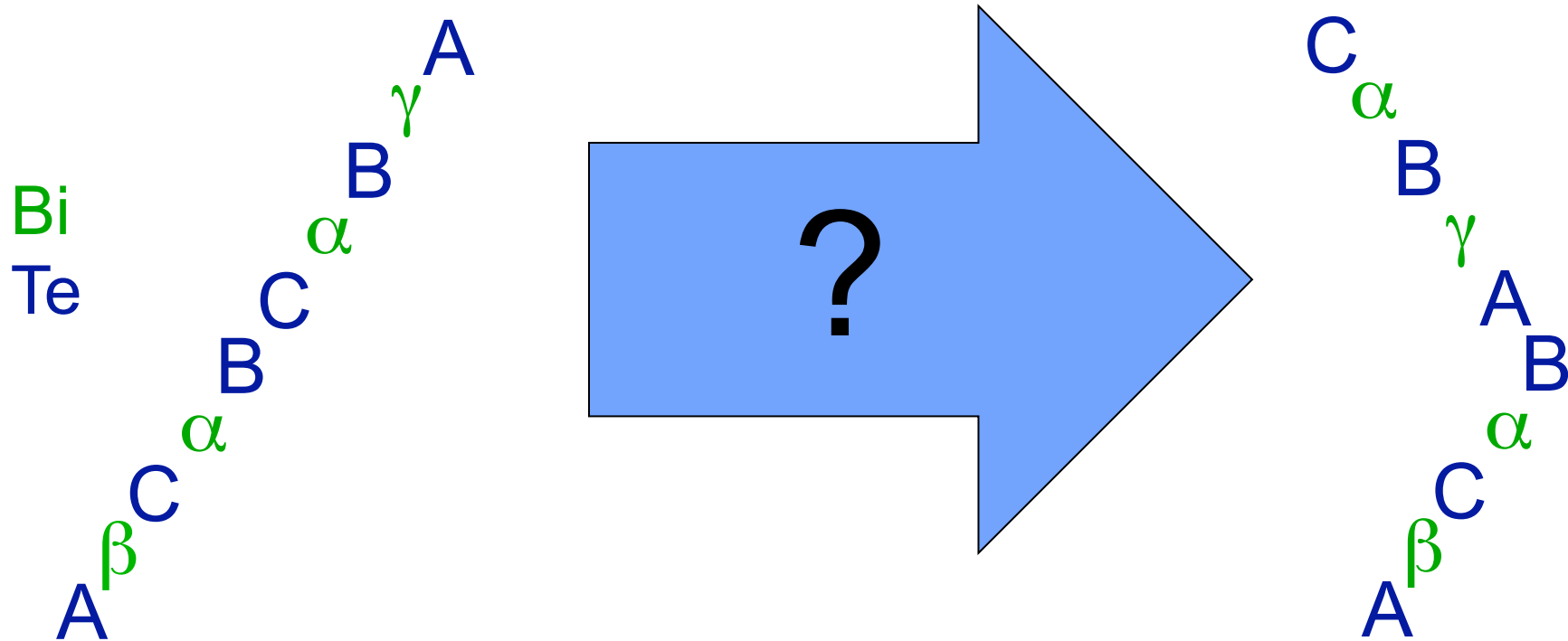


Lucadamo, Medlin, Talin, Yang, Kelly, Phil Mag 2005

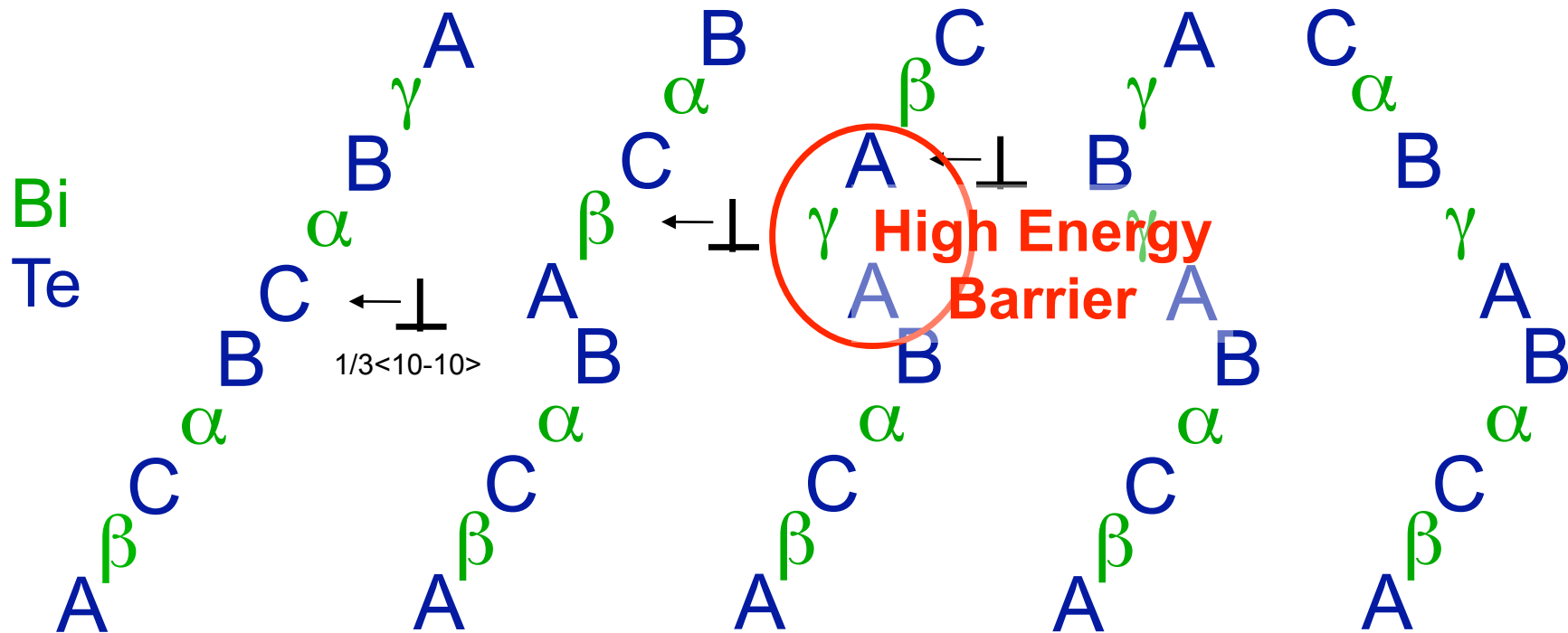
## Morphology Related to Dislocation Structure



# Sequential motion of twinning dislocations?



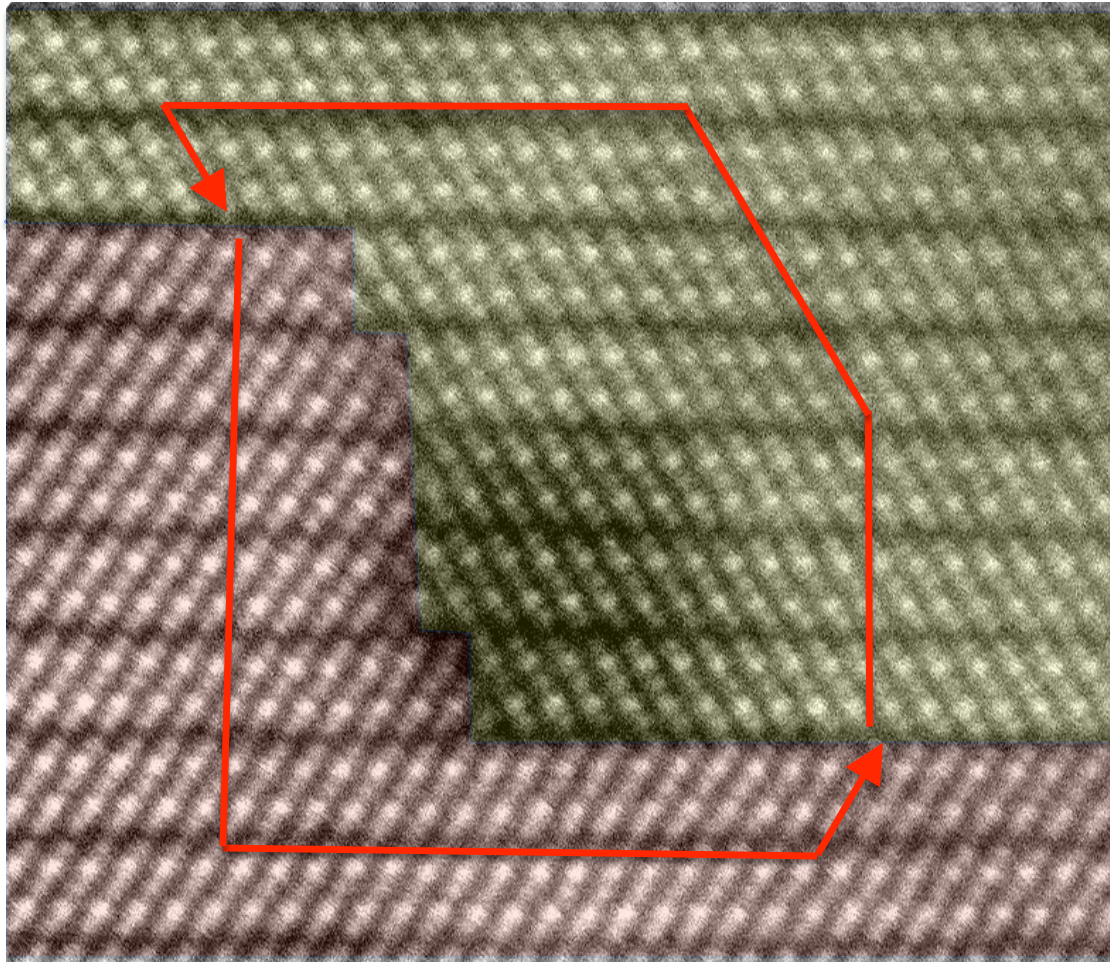
# Sequential motion of twinning dislocations?



- High energy barrier for Bi-terminated interface  
→ *independent motion of elementary twinning dislocations unlikely.*
- Alternative: Coordinated defect motion.  
Groupings of *unlike* twinning dislocations



# What is the dislocation content of the step?



$$\mathbf{b} = -(\mathbf{C}_\lambda + \mathbf{P}\mathbf{C}_\mu)$$

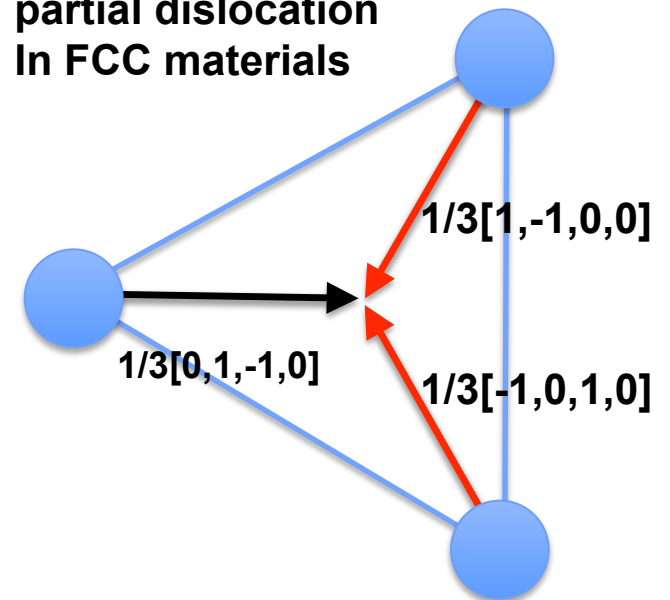
Upper circuit

Coordinate transformation

lower circuit

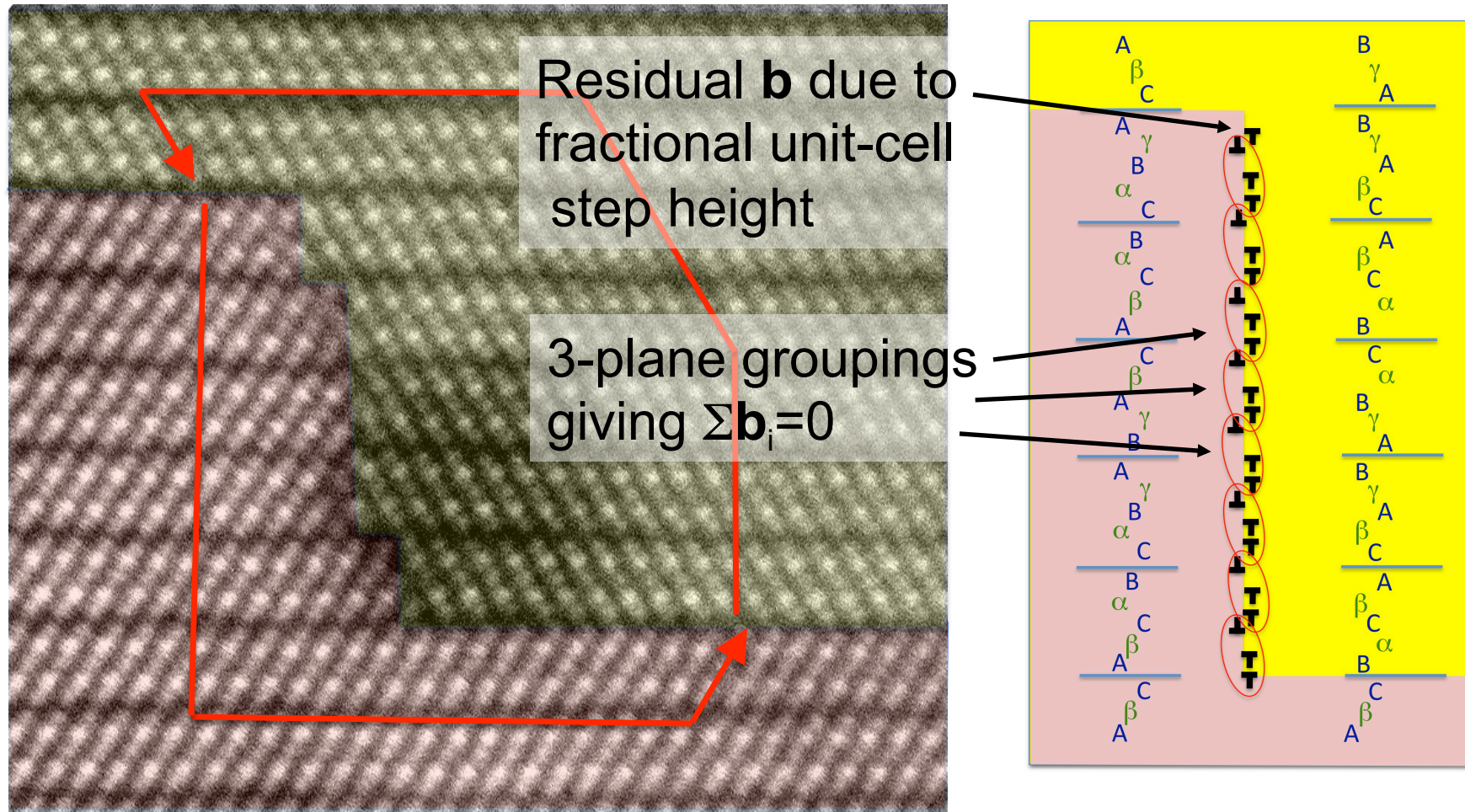
$$\mathbf{b} = \frac{1}{3}[-1, 0, 1, 0] \text{ or } \frac{1}{3}[1, -1, 0, 0]$$

Analogous to 30° Shockley partial dislocation  
In FCC materials





# What is the dislocation content of the step?



-Similar to  $\{112\}$  facets at annealing/growth twins in FCC materials

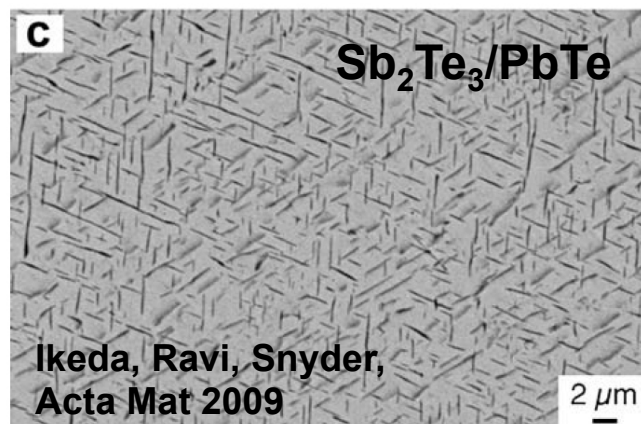
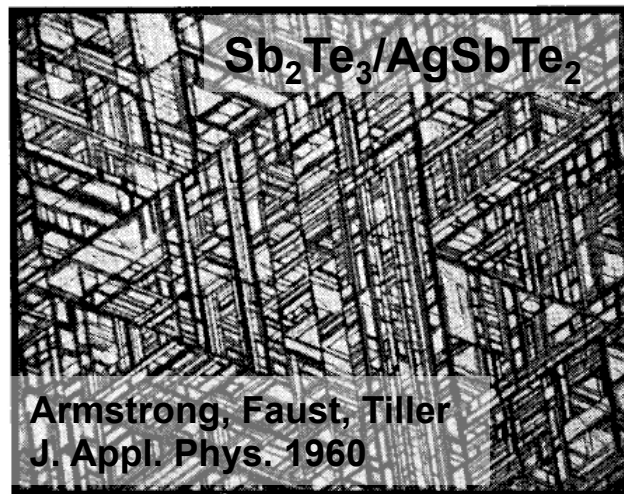
-Additional constraint: heterogeneity of chemical bonding.

-Dislocation models: route to generalizing to more complex boundaries

# Heterophase Boundaries

## *Rocksalt/Tetradymite Telluride Interfaces*

Widmanstätten plates



-Interest in forming thermoelectric nanocomposites of rock-salt and tetradymite tellurides:

- Possibility for well ordered interfaces.
- Transformations provide bulk route to synthesis.

*Ikeda, et al., Chem Mater. 2007*

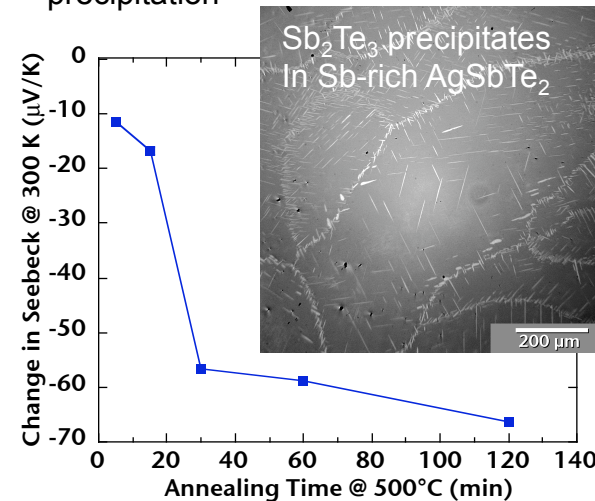
*Snyder and Toberer, Nature Materials 2008*

- $\text{AgSbTe}_2$ :

- Constituent of TAGS  $(\text{GeTe})_x(\text{AgSbTe}_2)_{1-x}$  and LAST  $(\text{PbTe})_x(\text{AgSbTe}_2)_{1-x}$   $zT \sim 1.8$

-High performance TE material:  $zT > 1.2$

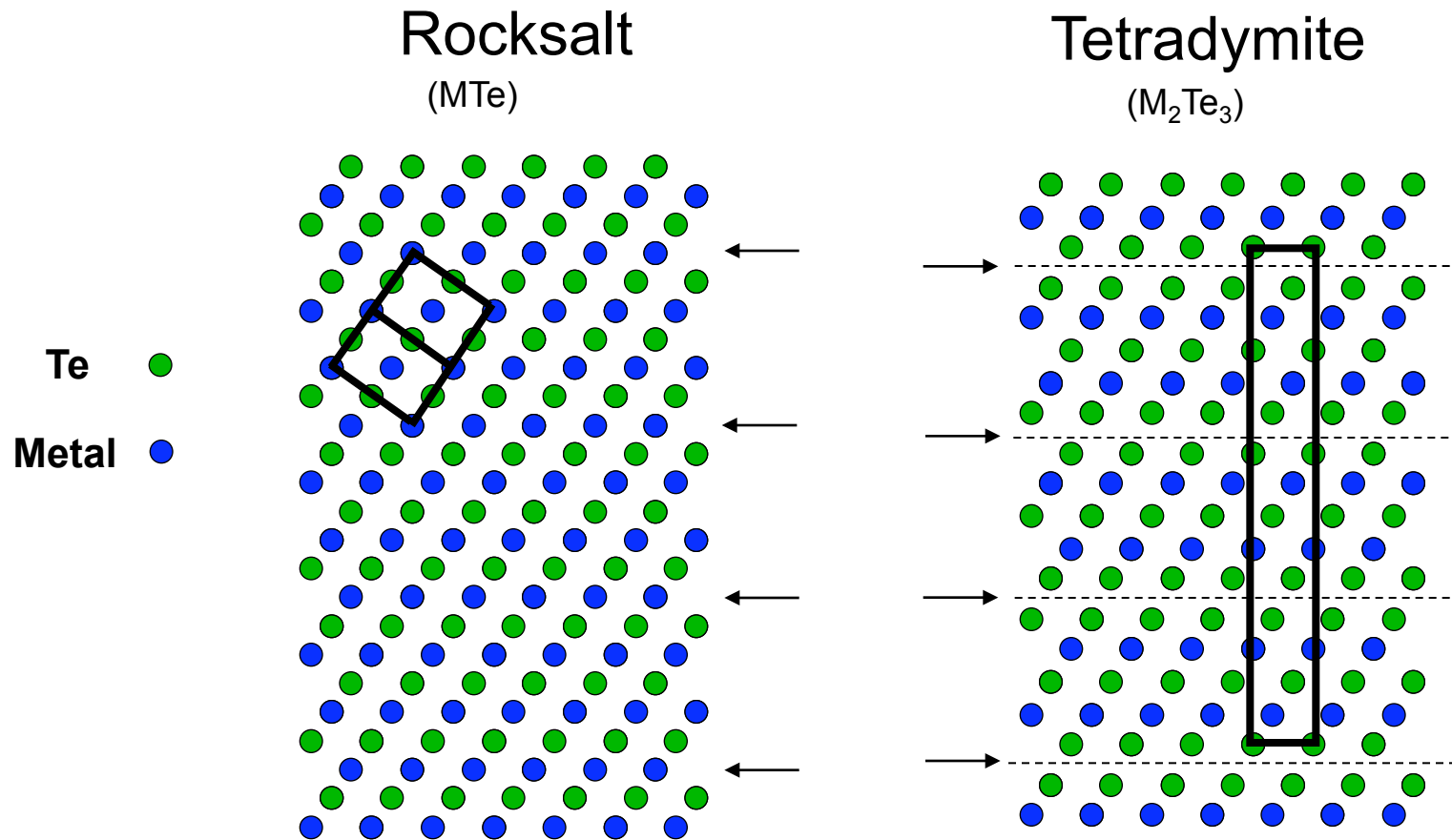
-Degradation of Seebeck coefficient with  $\text{Sb}_2\text{Te}_3$  precipitation



*Sharma, Sugar, Medlin J. Appl. Phys. 2010*



# Rocksalt and Tetradymite-structured Tellurides are Closely Related



- Remove metal plane every 6 layers
- Shear blocks by  $\frac{1}{3}\langle 10-10 \rangle$   
(or  $\frac{1}{6}\langle 112 \rangle$  relative to cubic coordinates)

# Transformation mechanism: *Motion of interfacial line defects*

Resolve  $\mathbf{b}$  into components  
normal and parallel to interface

$$\mathbf{b}_n = (a_{cub} - c_{hex} / 3\sqrt{3})[111]$$

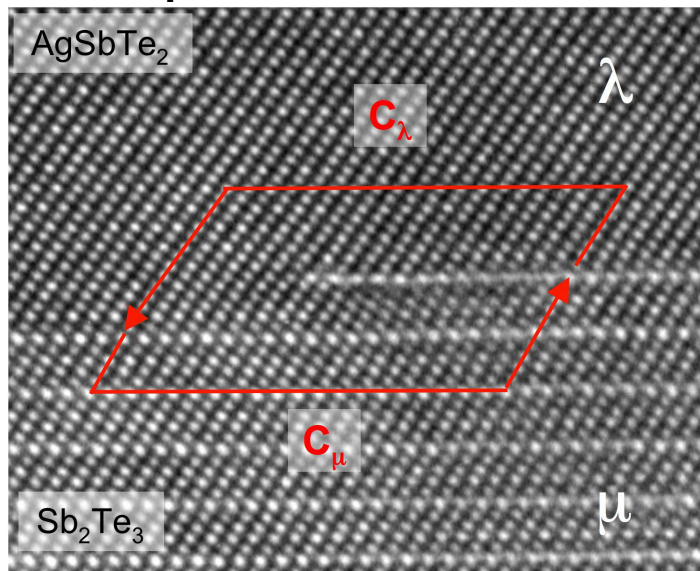
• *mismatch of step heights.*

•  $|\mathbf{b}_n| = 0.37 \text{ \AA}$

$$\mathbf{b}_{||} = \frac{a_{cub}}{6} [\bar{1}2\bar{1}]$$

• *Analogous to Shockley partial Dislocation*  
•  $|\mathbf{b}_{||}| = 2.48 \text{ \AA}$

Example: +6/+5 disconnection



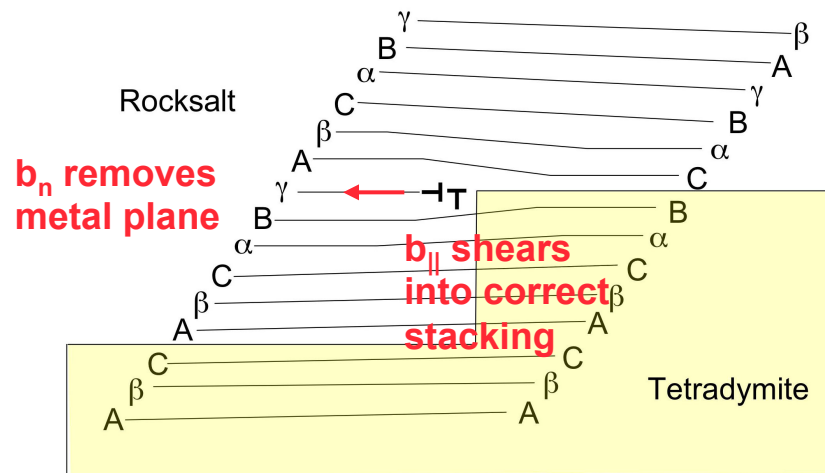
Circuit analysis to determine Burgers vector

Upper crystal circuit

Lower crystal circuit

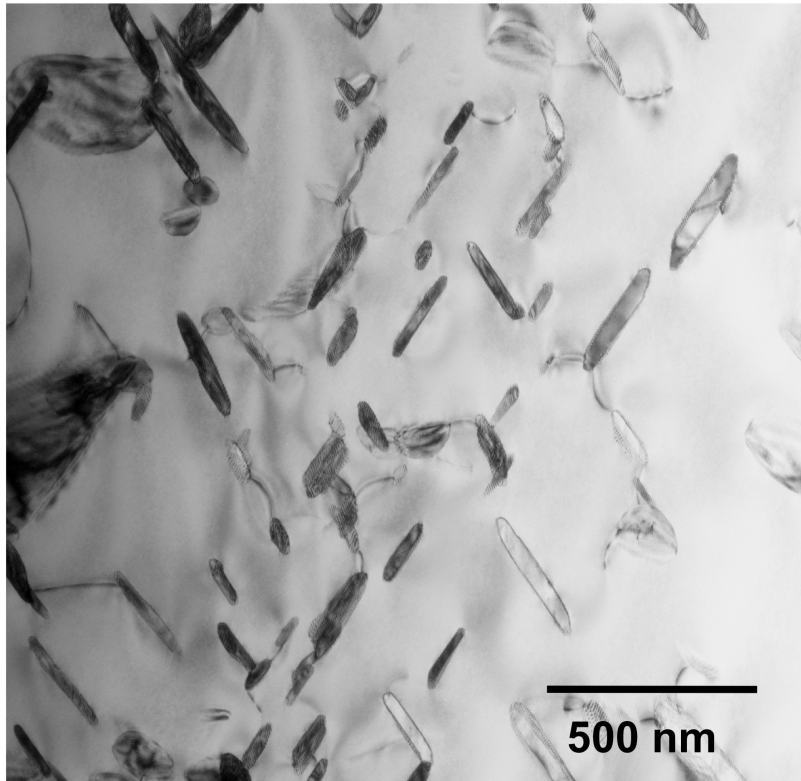
$$\mathbf{b} = -(\mathbf{C}_\lambda + \mathbf{P}\mathbf{C}_\mu)$$

Coordinate Transformation  
From Tetradymite to  
Rock-salt reference frame  
(coherent frame)



# A system with larger misfit: PbTe/Sb<sub>2</sub>Te<sub>3</sub>

Sb<sub>2</sub>Te<sub>3</sub> Precipitates in PbTe



AgSbTe<sub>2</sub>(111)/Sb<sub>2</sub>Te<sub>3</sub>(0001)

Misfit: +0.79%

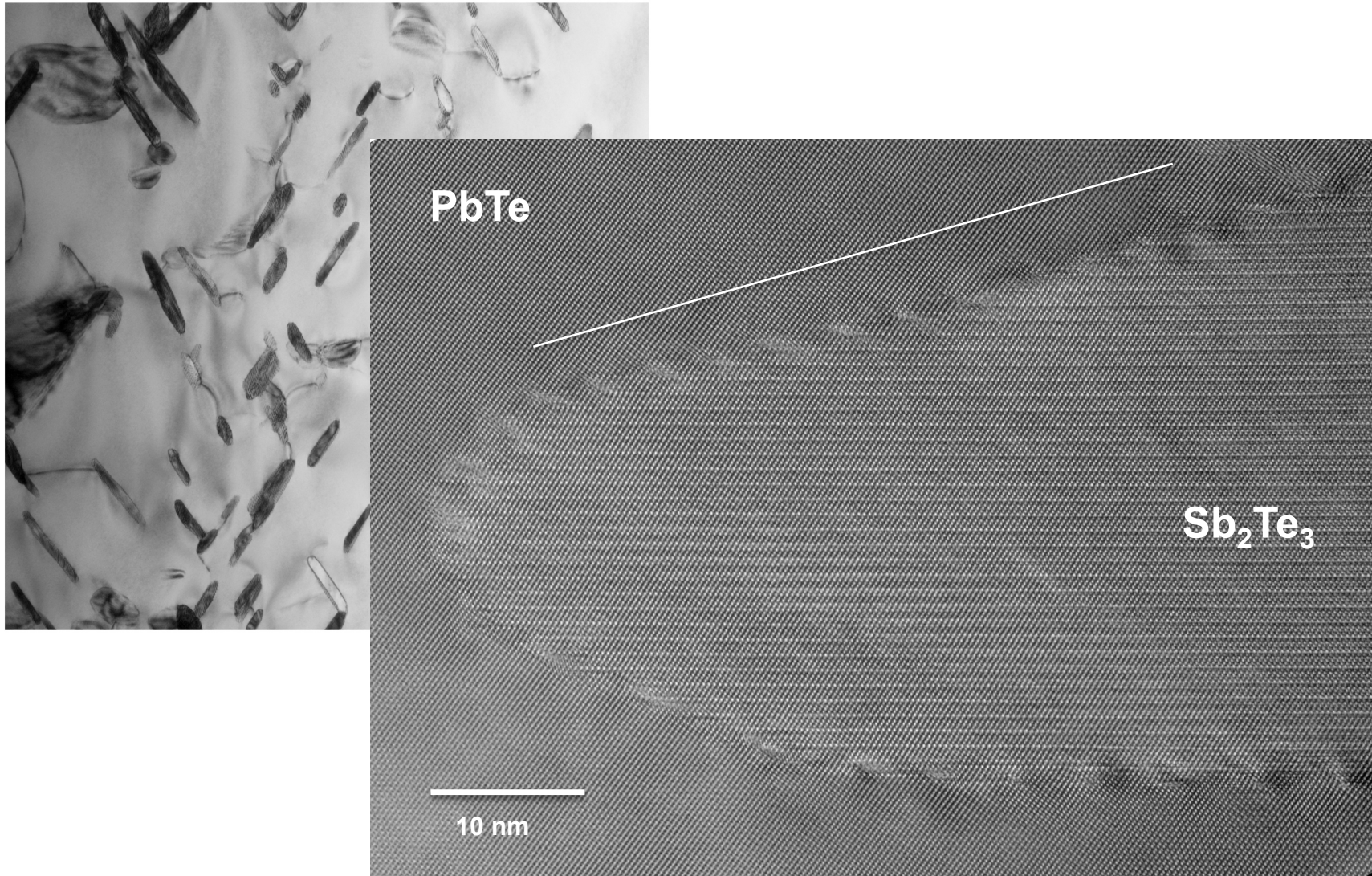
PbTe(111)/Sb<sub>2</sub>Te<sub>3</sub>(0001)

Misfit: +6.7%



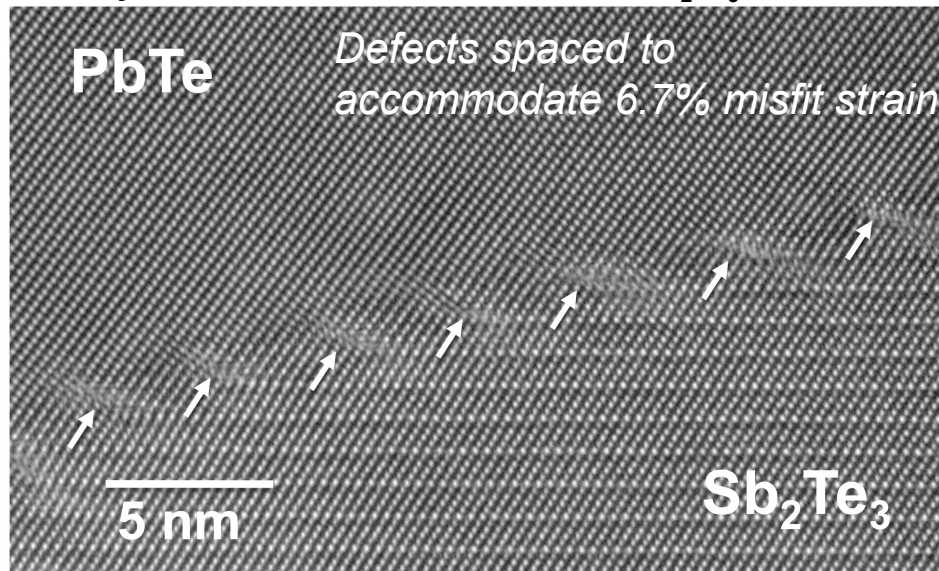
# A system with larger misfit: $\text{PbTe}/\text{Sb}_2\text{Te}_3$

$\text{Sb}_2\text{Te}_3$  Precipitates in  $\text{PbTe}$



# Interfacial defects: dual roles in transformation and strain accommodation

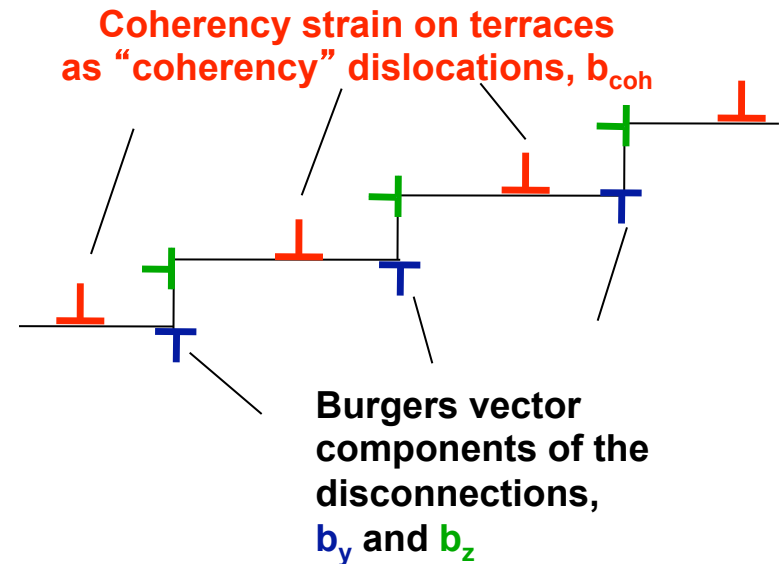
Array of +6/+5 disconnections at PbTe/Sb<sub>2</sub>Te<sub>3</sub> Interfaces



## Interface Geometry:

Inclination:  $\theta = 14.8^\circ$

Lattice rotation:  $\phi = 1.2^\circ \pm 0.5^\circ$



Project components on inclined interface plane  
Adjust spacing to cancel ( $\epsilon'_{yy} = 0$ )

$$-\epsilon = (b_y \tan \theta + b_z \tan^2 \theta) h^{-1} \quad \rightarrow \theta = 16.2^\circ$$

Pond, Celotto, Hirth, Acta Mater. 2003

Out of plane **b** components produce small rotation,

$$\phi = 2 \sin^{-1} [(b_z \cos \theta - b_y \sin \theta - \epsilon h \cos \theta) \sin \theta / 2h] \quad \rightarrow \phi = 1.7^\circ$$

• Local and long-range morphology of precipitates controlled by defect arrangements and connected to misfit strain relief.



# Summary

- Line defects provide “building blocks” for larger-length scale models for interface structure, properties, and behavior.
- Essential to understanding interface formation and stability in nanostructured bulk thermoelectrics.
- **Bi<sub>2</sub>Te<sub>3</sub> (0001) Twin boundary:**
  - Termination at Te(1)-Te(1) layer
    - Lowest energy structure from *ab initio* calculations
    - Structure confirmed with HAADF-STEM
  - Step Structure:
    - Analogies to FCC twins and defects.
    - chemical constraint due to energetics of interface termination
- **Mechanism for tetradymite plate growth by motion of interfacial defects:**
  - Defect plays two roles:
    - removes metal plane producing Te double-layer.
    - Shears layers into correct stacking.
  - Defect spacing consistent with accommodation of misfit strain.
    - Systematic departure from (111)/(0001) habit plane
  - Motion of defects unlikely to be independent;
    - instead must be coupled through strain interaction.*



# Related Publications

D.L. Medlin and N.Y.C. Yang, "Interfacial Step Structure at a (0001) Basal Twin in  $\text{Bi}_2\text{Te}_3$ ," *Journal of Electronic Materials* (in press) (2012). doi: 10.1007/s11664-011-1859-7

N.A. Heinz, T. Ikeda, G.J. Snyder, and D.L. Medlin, "Interfacial Disconnections at  $\text{Sb}_2\text{Te}_3$  Precipitates in PbTe: Mechanisms of Strain Accommodation and Phase Transformation at a Tetradymite/Rocksalt Telluride Interface," *Acta Materialia* 59 (20) (2011) 7724-7735. 10.1016/j.actamat.2011.08.043

J.D. Sugar and D.L. Medlin, "Solid-state Precipitation of Stable and Metastable Layered Compounds in Thermoelectric  $\text{AgSbTe}_2$ " *Journal of Materials Science* 46 (2011) 1668-1679. doi:10.1007/s10853-010-4984-4.

Y. Pei, J. Lensch-Falk, E.S. Toberer, D.L. Medlin, G.J. Snyder, "High Thermoelectric Performance in PbTe due to Large Nanoscale  $\text{Ag}_2\text{Te}$  Precipitates and La Doping" *Advanced Functional Materials* 2 (21) (2011) 241-249, doi: 10.1002/adfm.201000878.

D.L. Medlin, Q.M. Ramasse, C.D. Spataru, N.Y.C. Yang, "Structure of the (0001) Basal Twin Boundary in  $\text{Bi}_2\text{Te}_3$ " *Journal of Applied Physics* 108 (2010) 043517. doi:10.1063/1.3457902.

J. L. Lensch-Falk, J.D. Sugar, M.A. Hekmaty, D.L. Medlin, "Morphological Evolution of  $\text{Ag}_2\text{Te}$  Precipitates in Thermoelectric PbTe" *Journal of Alloys and Compounds* 504 (2010) 37-44, doi:10.1016/j.jallcom.2010.05.054.

P.A. Sharma, J.D. Sugar, D.L. Medlin "Influence of Nanostructuring and Heterogeneous Nucleation on the Thermoelectric Figure of Merit in  $\text{AgSbTe}_2$ " *Journal of Applied Physics* 107 (2010) 113716. doi: 10.1063/1.3446094.

D.L. Medlin and J.D. Sugar, "Interfacial Defect Structure at  $\text{Sb}_2\text{Te}_3$  Precipitates in the Thermoelectric Compound  $\text{AgSbTe}_2$ " *Scripta Materialia*. 62 (2010) 379-382 doi:10.1016/j.scriptamat.2009.11.028

J.D. Sugar and D.L. Medlin, "Precipitation of  $\text{Ag}_2\text{Te}$  in the Thermoelectric Material  $\text{AgSbTe}_2$ ", *Journal of Alloys and Compounds* 478 (2009) 75-82. doi:10.1016/j.jallcom.2008.11.054.

D.L. Medlin and G.J. Snyder, "Interfaces in Bulk Thermoelectric Materials", *Current Opinion in Colloid and Interface Science* 14 (2009) 226-235. doi: 10.1016/j.cocis.2009.05.001.

# Collaborators for this work

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Jessica Lensch-Falk  
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Catalin Spataru  
Michelle Hekmaty  
Mark Homer

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Teruyuki Ikeda  
G. Jeffrey Snyder

## **LBNL:**

Quentin Ramasse  
Currently: SuperSTEM, Daresbury, UK

## ***Special thanks to:***

- LBNL: User program, National Center for Electron Microscopy
- LLNL: John Bradley, for use for LLNL's Titan 80/300 instrument
- UCD: Z. Zhang and E. Lavernia, for assistance with SPS processing

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